

Introduction to Organic Electronics

Organic Thin-film Transistors

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History of organic thin-film transistors (OTFTs)

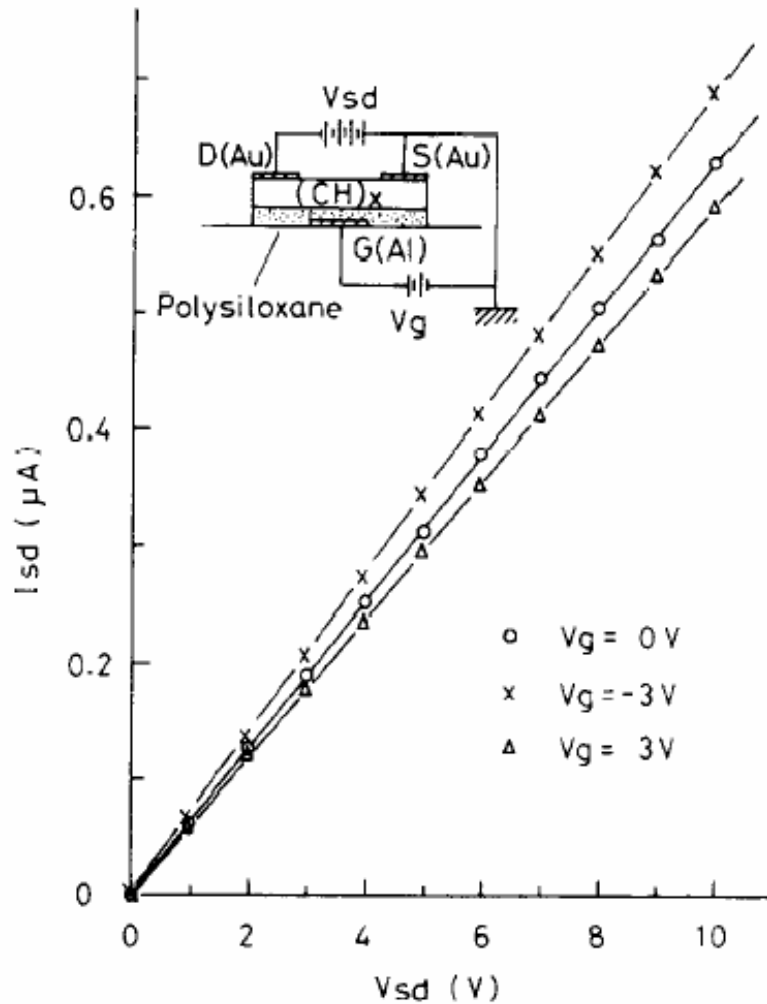
1970's --- the first world energy crisis launched interests in organic semiconductor (for solar cells)

1983 --- the first OTFT based on polyacetylene
Air-sensitive, low mobility ($10^{-5} \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$)

1990 --- the OTFT based on sioxithiophene
mobility $\sim 0.1 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$ (comparable with a-Si:H)

1997 --- the OTFT based on pentacene
mobility $\sim 1 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$

The First OTFTs

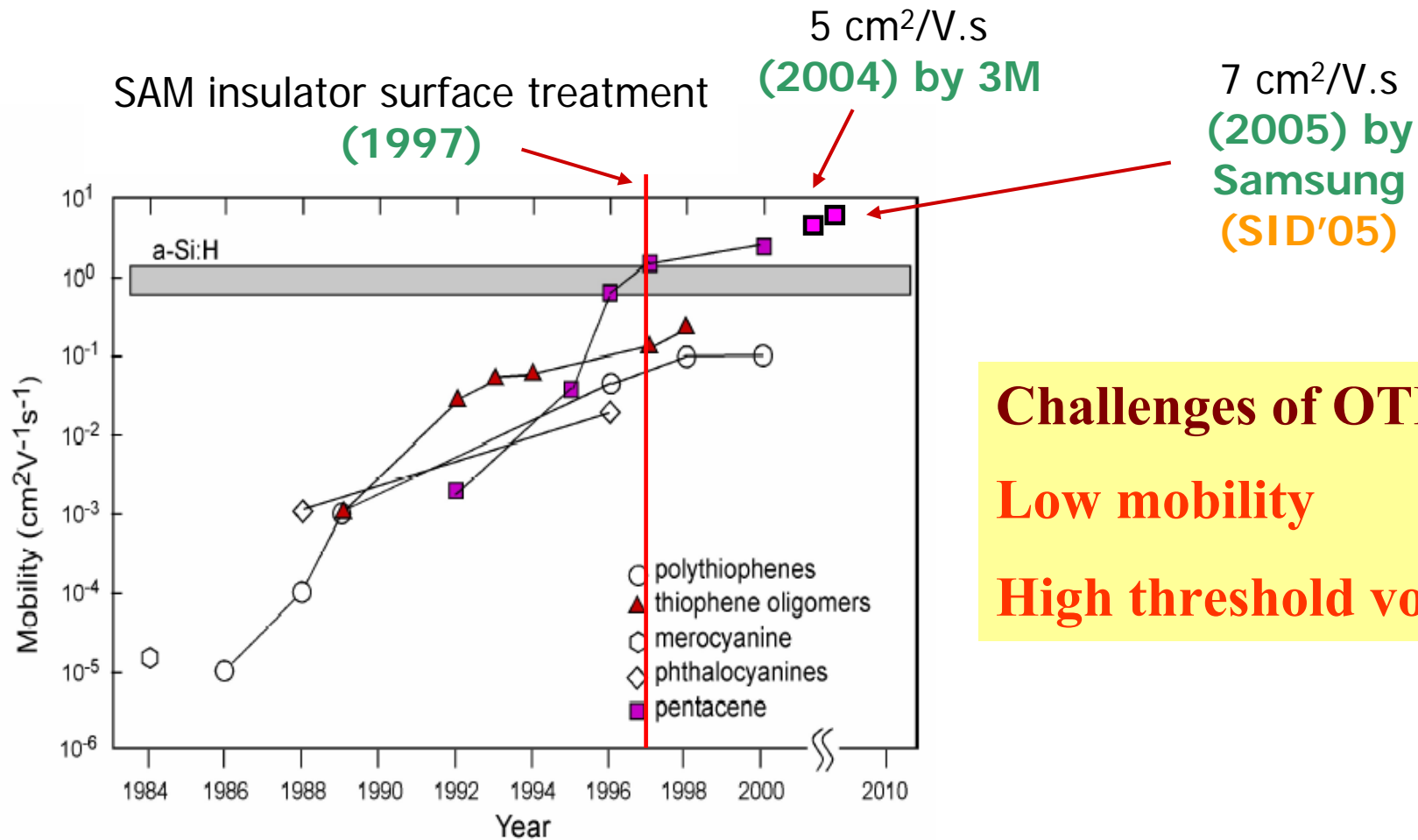


polyacetylene/polysiloxane

Very low mobility (10^{-5})

Nonexistence of pinch-off

Evolution of OTFT hole mobility



Challenges of OTFTs

Low mobility

High threshold voltage

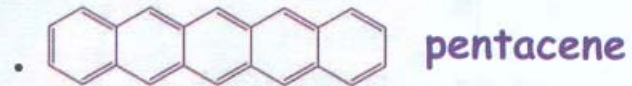
C. D. Dimitrakopoulos, and P. R. L. Malenfant Adv. Mater., 14, 99 (2002)

Organic thin-film transistors (OTFTs)

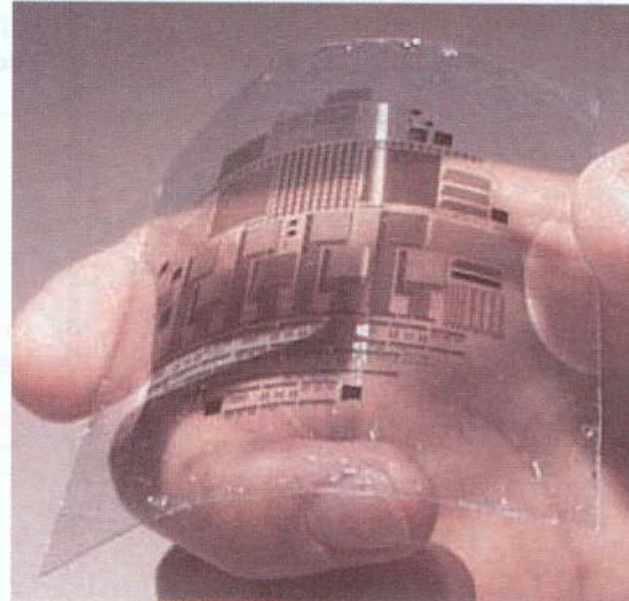
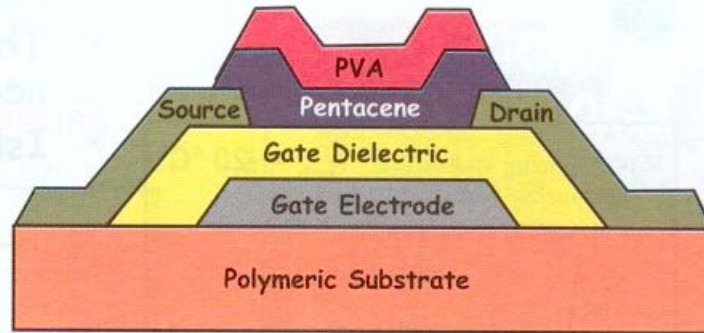
- Low temperature processing allows arbitrary substrates and flexible processing

- Simple device fabrication

Pentacene:

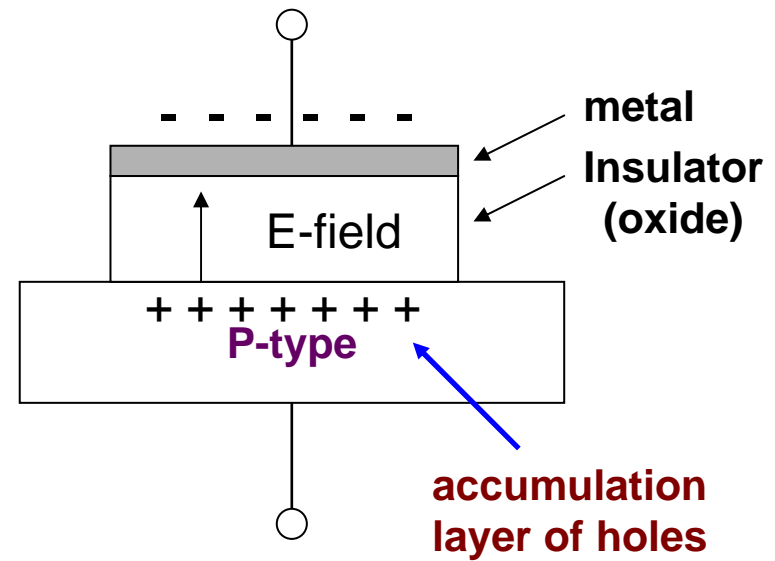
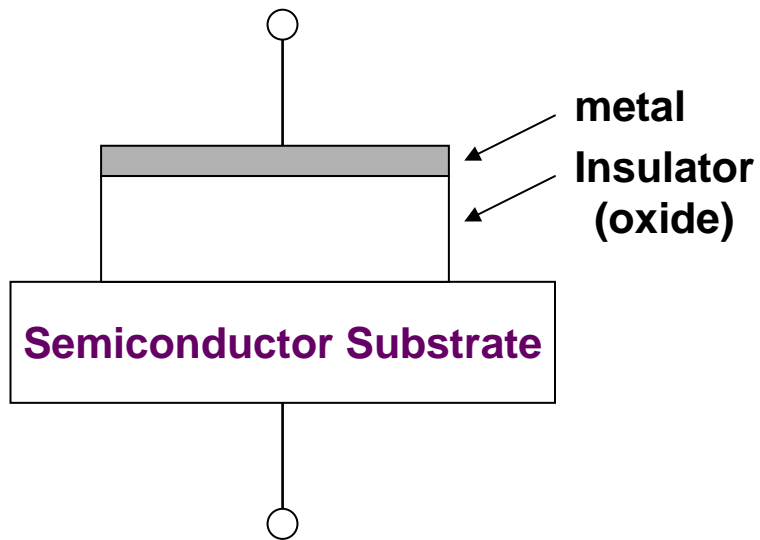
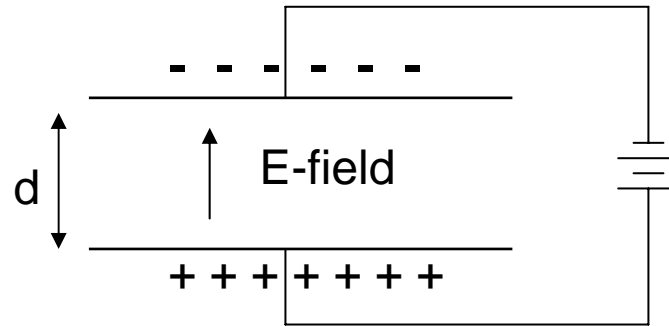


- Small molecule organic semiconductor
- Thin film mobility $> 3 \text{ cm}^2/\text{V}\cdot\text{s}$, $\sim 1 \text{ cm}^2/\text{V}\cdot\text{s}$ typical
- Simple low-temperature vacuum deposition
- Strong tendency to form well-ordered molecular crystals



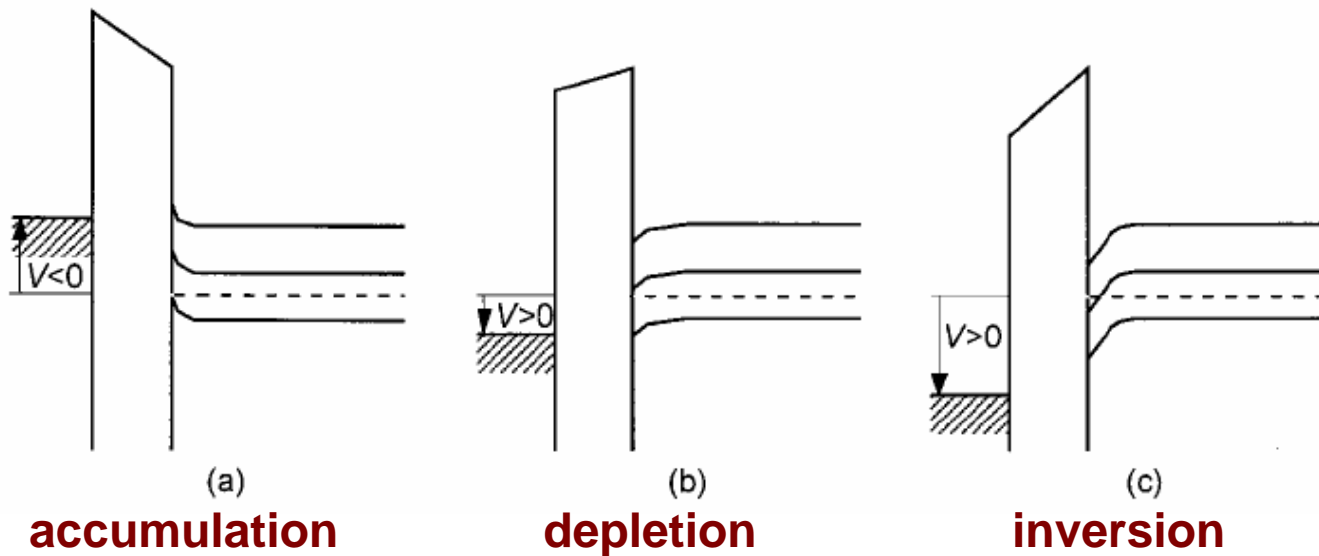
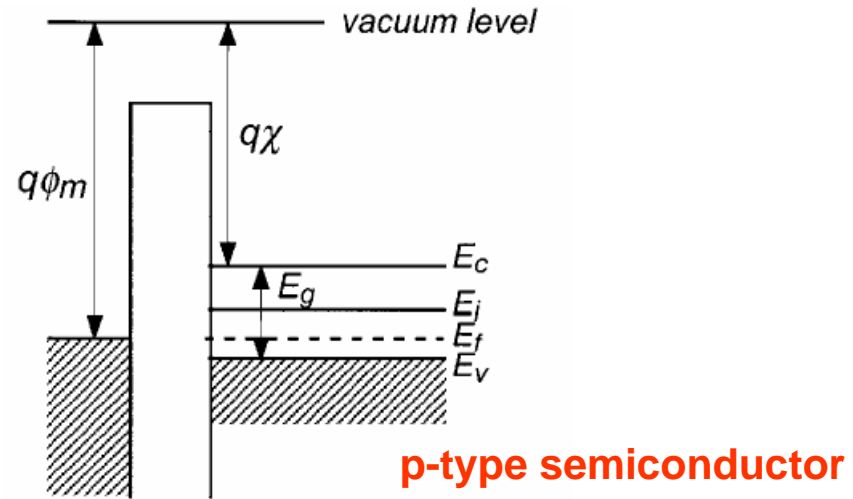
Metal-Insulator-Semiconductor (MIS)

$$C = \epsilon A/d$$



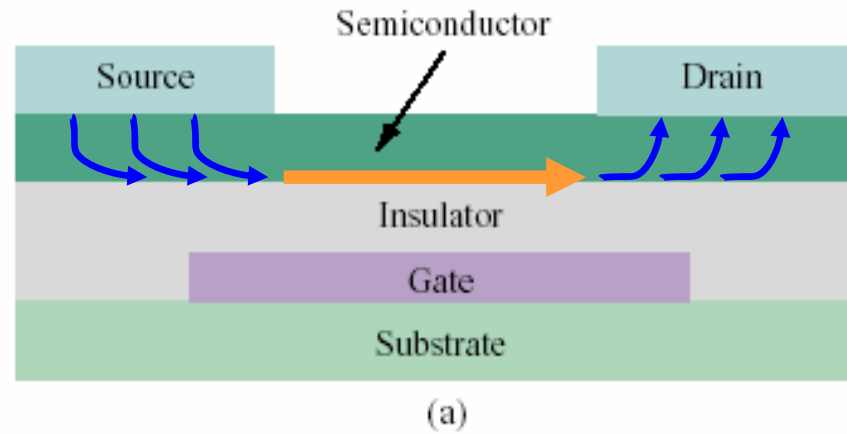
Metal-Insulator-Semiconductor Junction

Ideal MIS band diagram

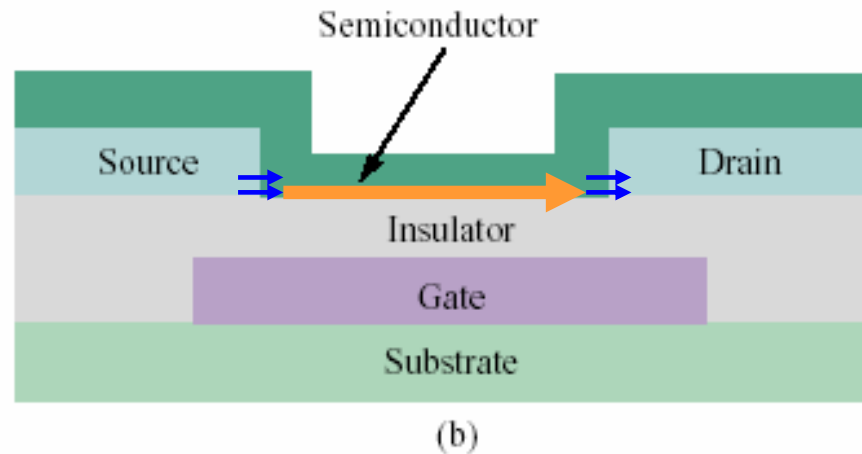


OTFT device configurations

Staggered or
Top-contact (TC) Device

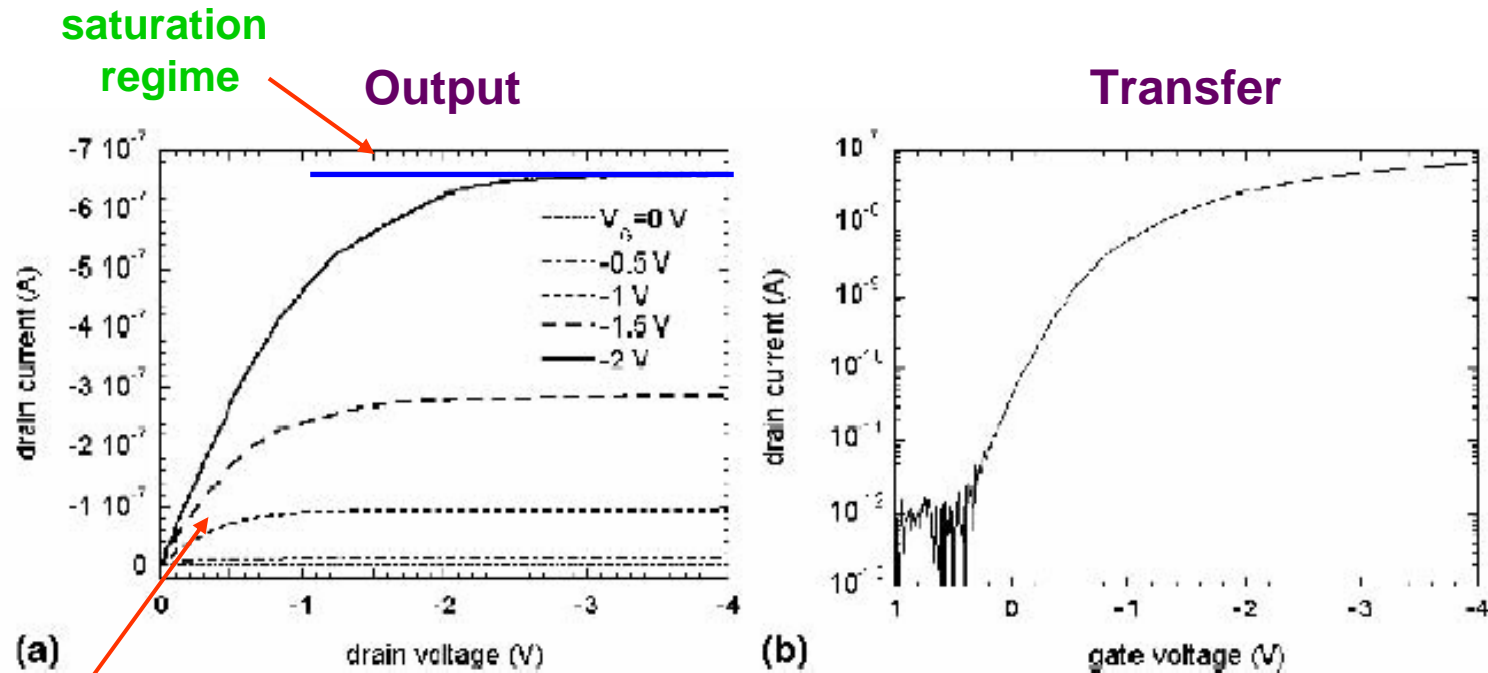


Coplanar or
Bottom-contact (BC) device



TC OTFT R_c < BC TFT R_c

Typical OTFT I-V characteristics



Ohmic's law

When the drain voltage is compared to gate voltage, the voltage drop at drain Contact falls to zero and the conducting channel is pinched off

3M's high mobility OTFTs

3M

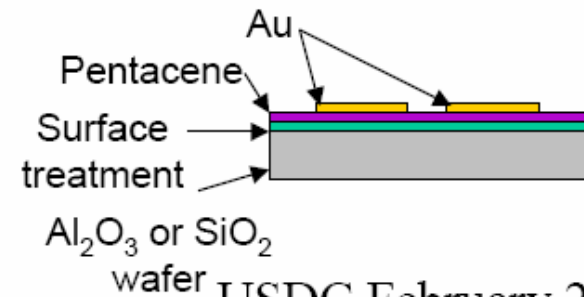
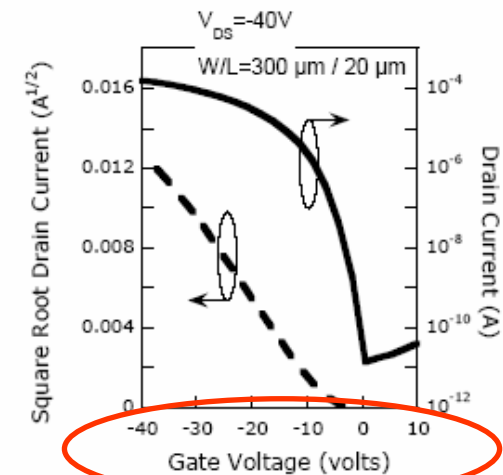
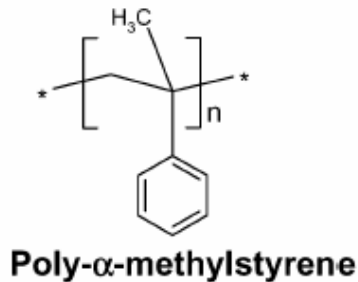
High Mobility Pentacene Devices

- Surface treatments
- Controlled deposition conditions
- Top contact construction

Average μ - pentacene on Al_2O_3

Without surface treatment
 $\sim 1 \text{ cm}^2/\text{Vs}$

With (AMS) surface treatment
 $\sim 5 \text{ cm}^2/\text{Vs}$



3M Corporate Research Materials Lab

USDC February 2004

Charge Transport

For conventional semiconductor,

Delocalized states, and is limited by the scattering of carriers,
Mainly phonons

$$T \downarrow \longrightarrow \mu \uparrow$$

$$\mu = a T^{-n}$$

(n ~ 1)

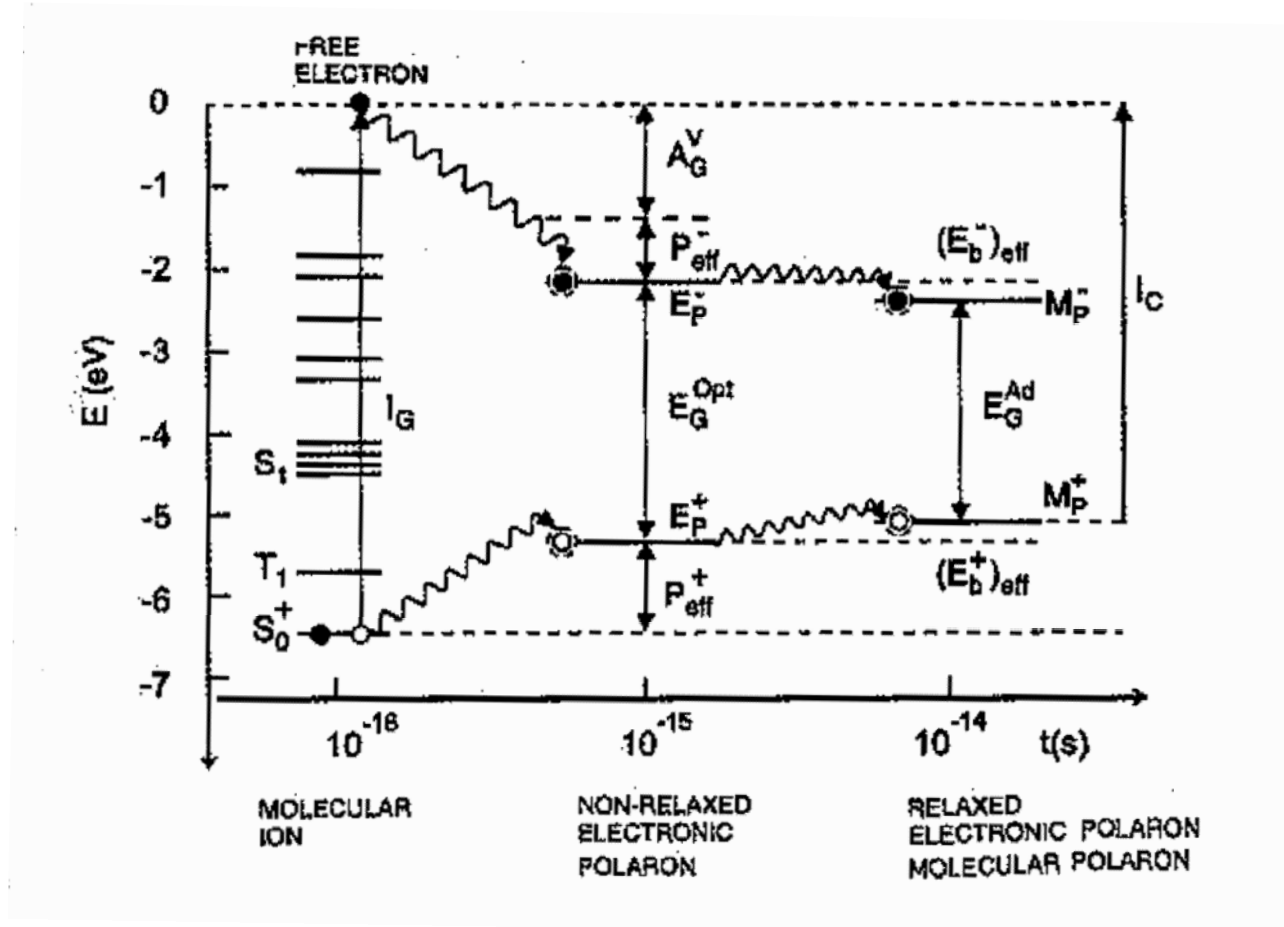
For Organic semiconductor,

Hopping between localized states, phonon assisted
(the mean free path of carriers < atomic distance)

$$T \uparrow \longrightarrow \mu \uparrow$$

$$\mu = \mu_0 \exp -(E_a/kT)$$

Electronic Polarons vs Molecular Polarons



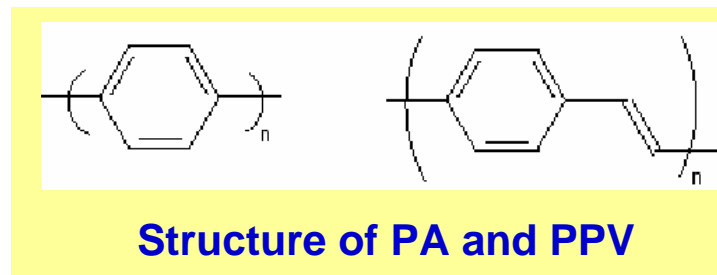
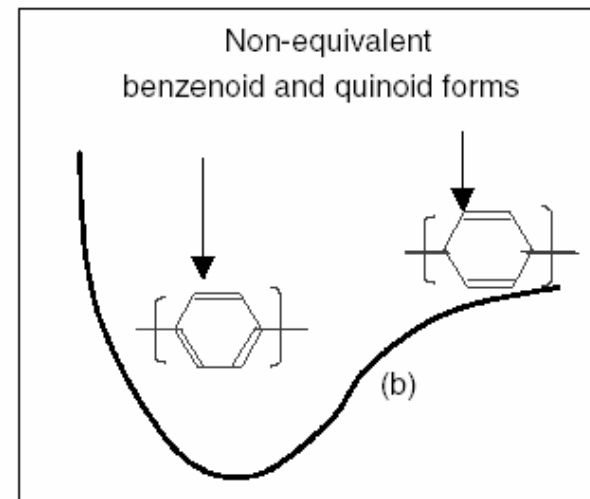
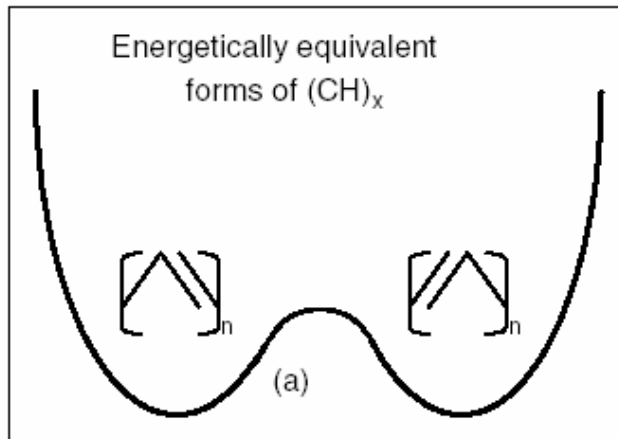
E_p^- & E_p^+ : **non-relaxed electronic polaron** states

P_{eff}^- & P_{eff}^+ : effective electronic polarization energies

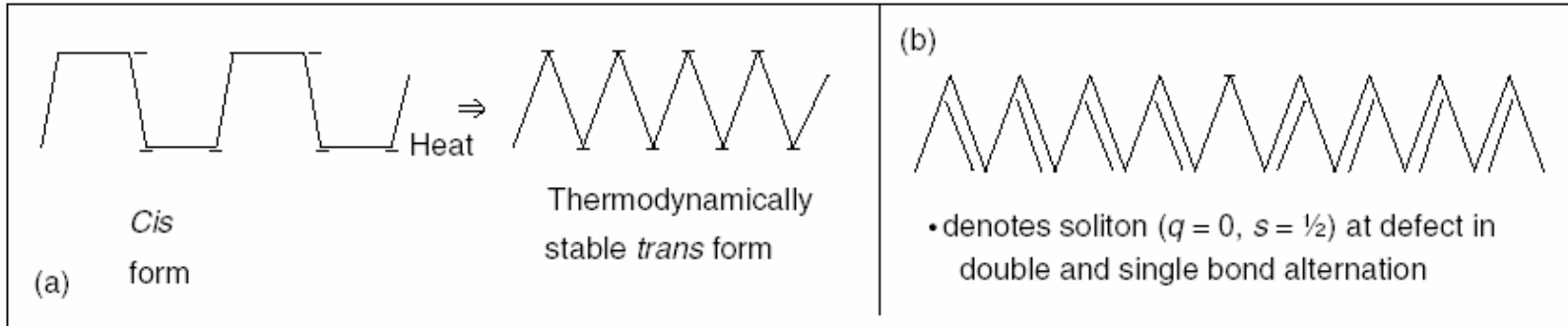
M_p^- & M_p^+ : **molecular polaron** conductivity levels

$(E^-)_{eff}$ & $(E^+)_{eff}$: effective formation energies of a molecular polaron
(due to vibronic polarization)

Degenerate PA and non-degenerate PPP

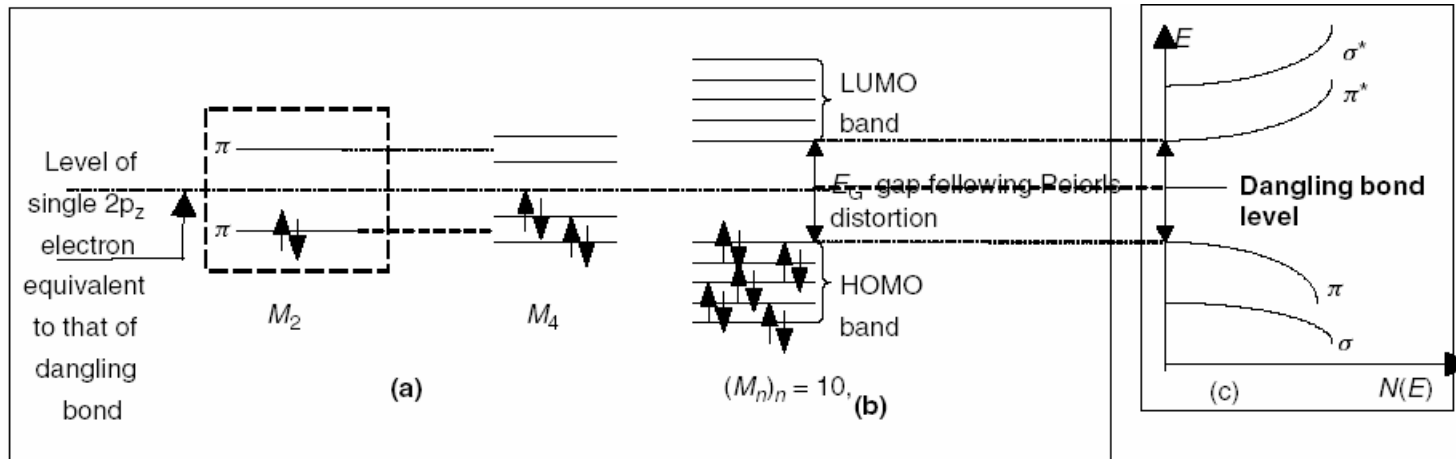


Soliton defect and energy level



Soliton defect in PA

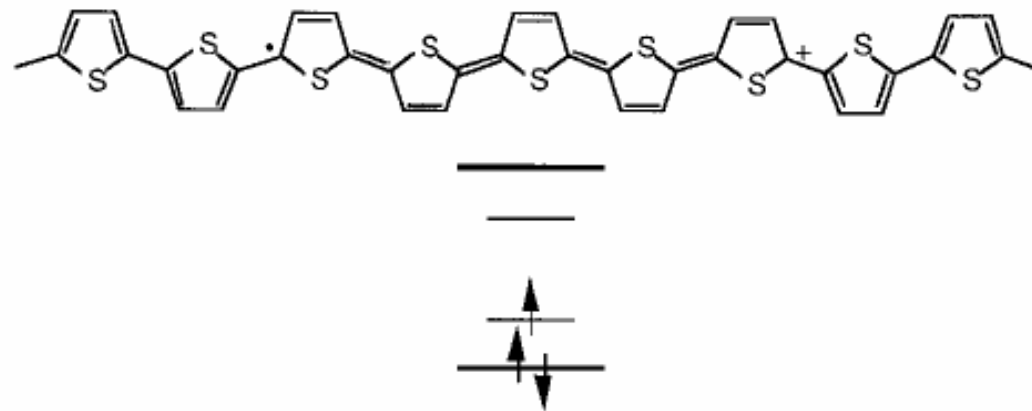
**Soliton: structure defect, a stable free radical
can propagate along the chain, may not carry any charge itself**



The small polaron

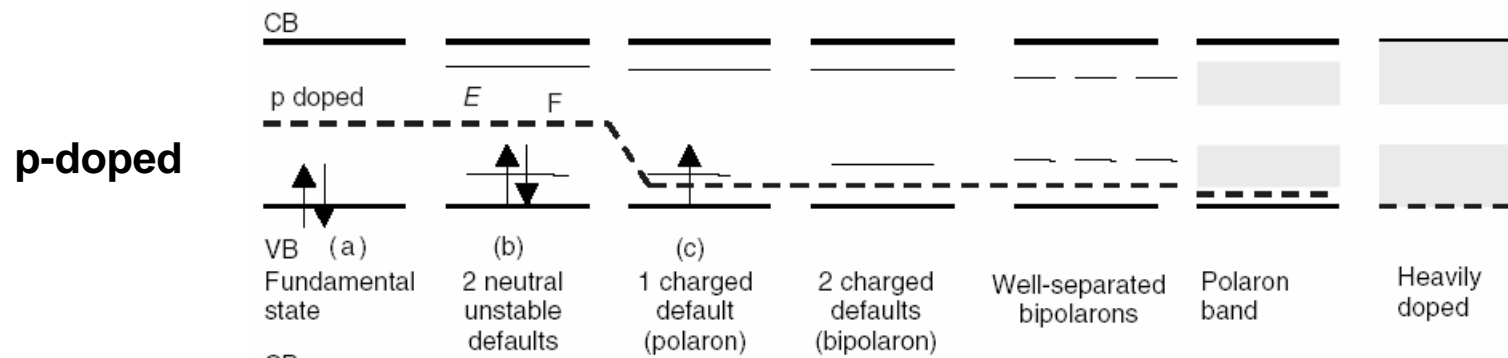
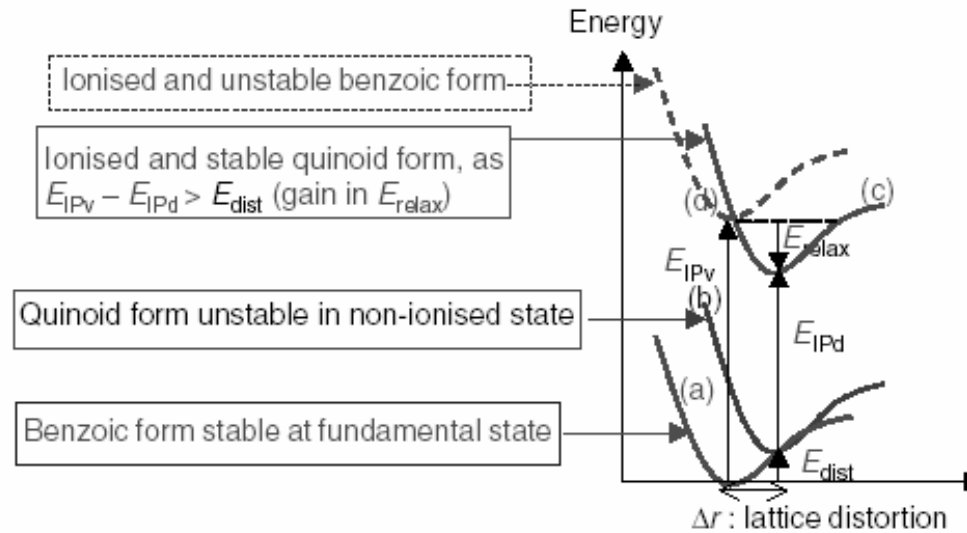
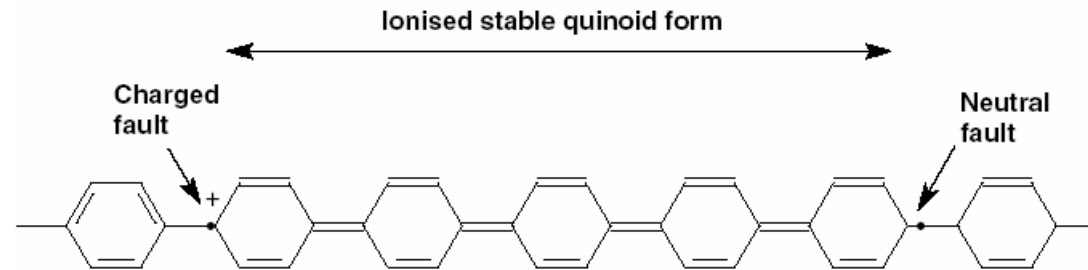
A polaron results from the deformation of the conjugated chain under the action of the charge.

The charge is “self-trapped” by the deformation it induced in the chain.



A polaron in polythiophene

Charge storage in conjugated polymers



The small polaron model : Holstein's model

The total energy : (three terms)

The lattice energy E_L ,

$$E_L = \sum_{n=1}^N \frac{1}{2M} \left(\hbar \frac{\partial}{\partial u_n} \right)^2 + \frac{1}{2} M \omega_0^2 u_n^2$$

N harmonic oscillators
vibrates at frequency, ω_0

The energy dispersion of the electron,

$$E_k = E_0 - 2J \cos(ka)$$

(J : the electron transfer energy)

The electron-lattice coupling

$$\varepsilon_n = -A u_n$$

Important parameter : polaron binding energy $E_b := A^2/(2M\omega_0^2)$

“Small” polaron : electronic bandwidth $2J < E_b$, (perturbation)

High temperature limit :

$$\mu = \sqrt{\frac{\pi}{2}} \frac{ea^2}{\hbar} \frac{J^2}{\sqrt{E_b}} (kT)^{-3/2} \exp\left(-\frac{E_b}{2kT}\right)$$

$\sim 1 \text{ cm}^2\text{V}^{-1}\text{s}^{-1} (ea^2/\hbar)$

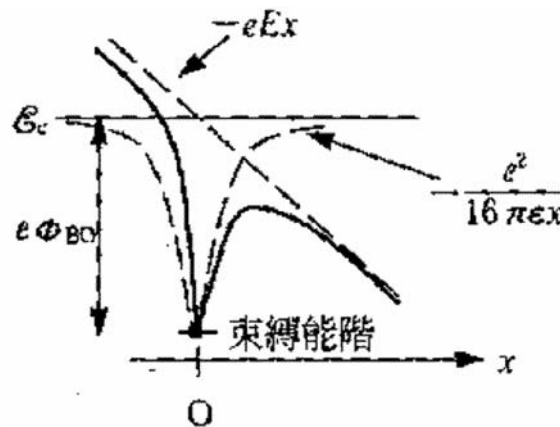
Field-Dependent Mobility

The mobility of organic materials become field-dependent at high field ($\sim 10^5$ V/cm) (a general feature of organic devices)

Poole-Frenkel mechanism,

the coulombic potential near the localized levels is modified by the applied field in such a way as to increase the tunnel transfer rate between sites.

$$\mu(F) = \mu(0)\exp\left(\frac{q}{kT}\beta\sqrt{F}\right)$$



Multiple Trapping and Release (MTR)

A narrow delocalized band with a high concentration of localized traps
(widely used in a-Si)

Assumptions:

1. Trap probability ~ 1
2. The release of trapped carriers is controlled by a thermal activated process.

$$\mu_D = \mu_0 \alpha \exp\left(-\frac{E_t}{kT}\right)$$

single trap

a single trapping level E_t

α is the ratio of effective density of states between the trap level and the delocalized band edge

Multiple Trapping and Release (MTR)

Gate-bias-dependent mobility

At low gate bias, field-induced charges go to localized levels

→ low mobility

At high gate bias, more localized levels are filled, and Fermi level approach the delocalized band, increasing mobility

However, at high field region, the mobility is still too low to correspond to transport in delocalized states

Two general types of organic materials

Two general cases

“Low” mobility materials

transport via hopping

typical mobility $10^{-6} \sim 10^{-1} \text{ cm}^2\text{V}^{-1}\text{sec}^{-1}$

common for polymeric or disorder organic semiconductors

“high” mobility materials

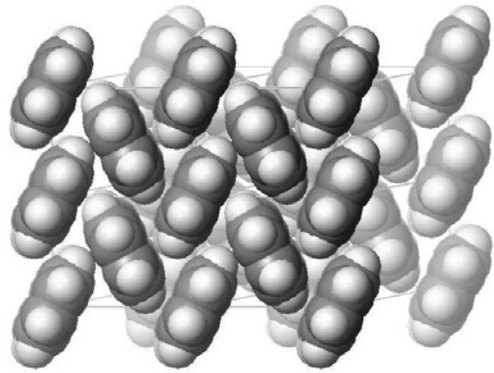
transport via narrow band transport

typical mobility $10^{-1} \sim 10 \text{ cm}^2\text{V}^{-1}\text{sec}^{-1}$

$\mu(T)$ depends on details (traps, doping, bandwidth, etc.)

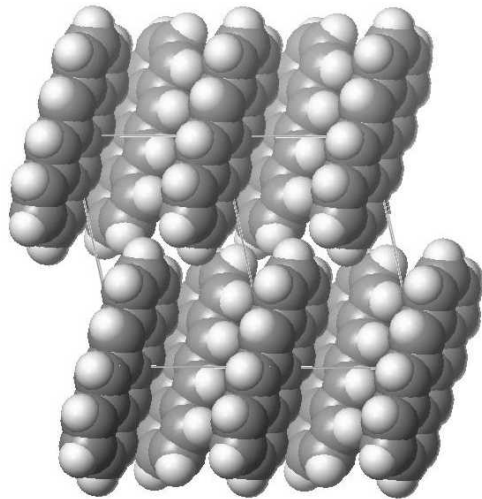
common for small molecular crystal organic semiconductors

Crystal Structure - Anisotropy



Herringbone-Structure

van-der-Waals (attraction)
Pauli-Principle (repulsion)



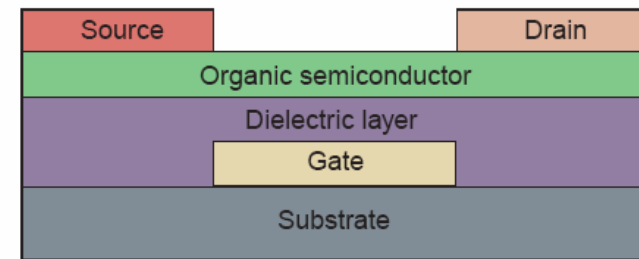
Layered Semiconductor

Tetracene

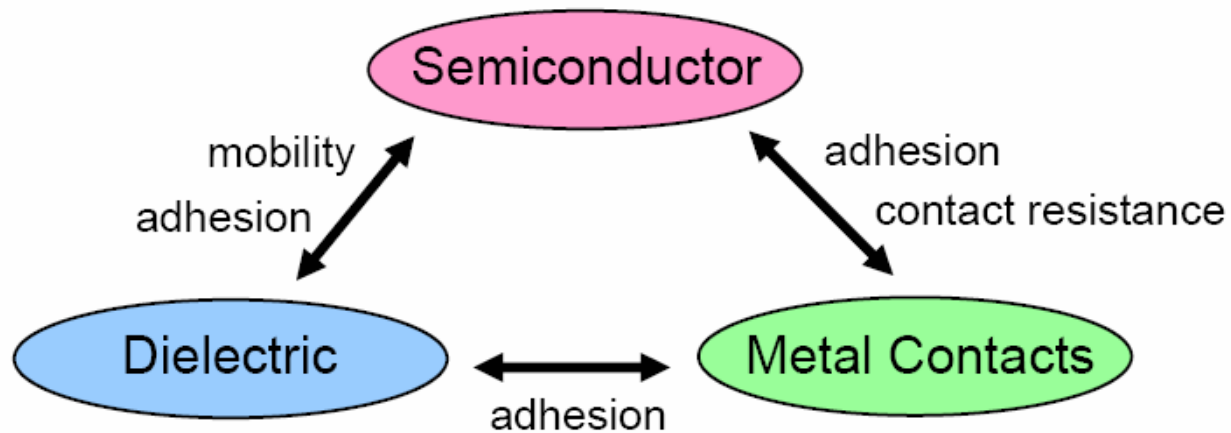
Anisotropy		
	$\sigma_{\text{par}} / \sigma_{\text{per}}$	$\sigma_{\text{par1}} / \sigma_{\text{par2}}$
Thiophenes	~70	~1.5
Acenes	~3	~1.5

Critical Issues of OTFTs

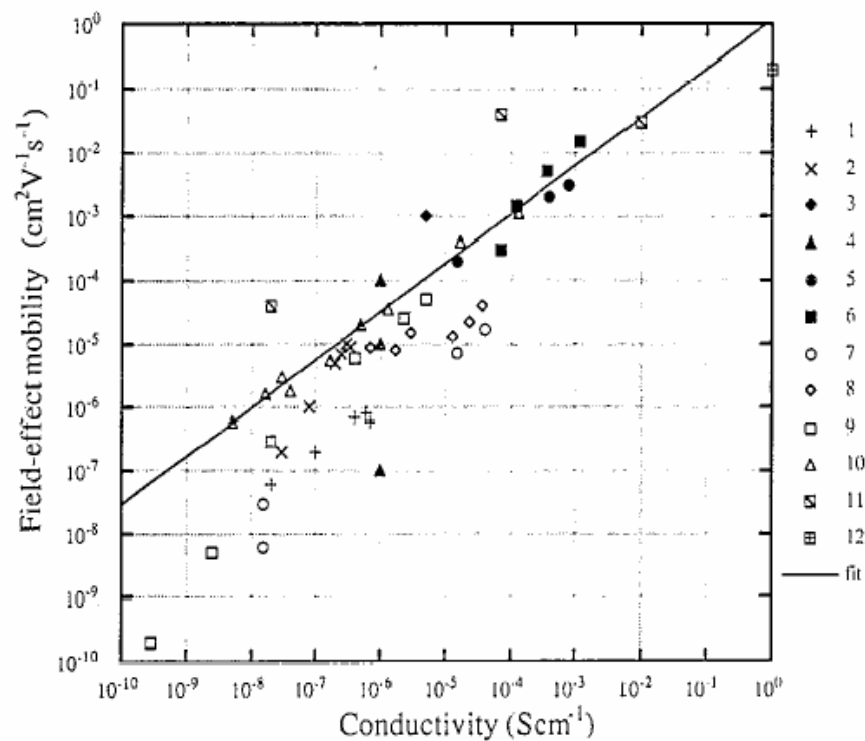
- Mobility
- On/Off Ratio
- Contact Resistance
- Dielectric Compatibility
- (Solvent Compatibility)



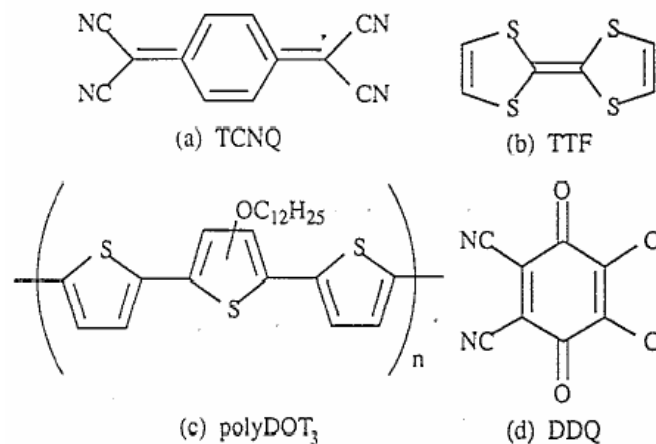
Top contact



Conductivity vs Mobility in doped amorphous organic semiconductor

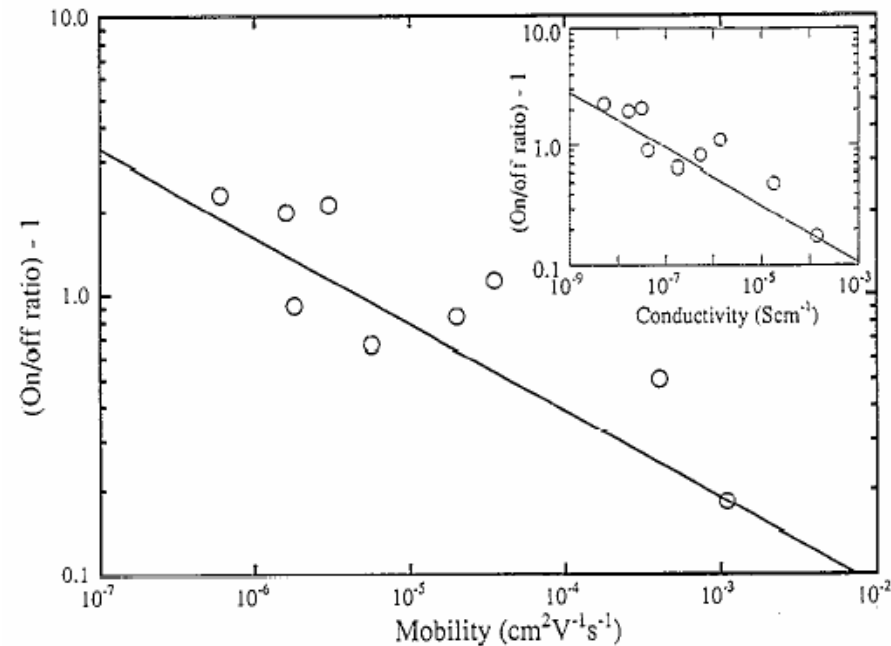


$$\mu \sim \sigma^\delta \quad (\delta = 0.76)$$



Data set	Material	Deposition technique	Ref.
1	PHT AA	Langmuir-Blodgett	[14]
2	QT AA	Langmuir-Blodgett	[14]
3	ScPc2	vacuum-deposited	[15]
4	P3HT	spin-coated	[16]
5	Pc2Lu	vacuum-deposited	[17]
6	Pc2Tm	vacuum-deposited	[17]
7	polyalkylthiophene	spin-coated	[18]
8	polyalkylthiophene	spin-coated	[39]
9	TCNQ	vacuum-deposited	[23]
10	polyDOT ₃	spin-coated	this work
11	C ₆₀	vacuum-deposited	[19]
12	(N-octa')-Ni(dmit) ₂	Langmuir-Blodgett	[20]

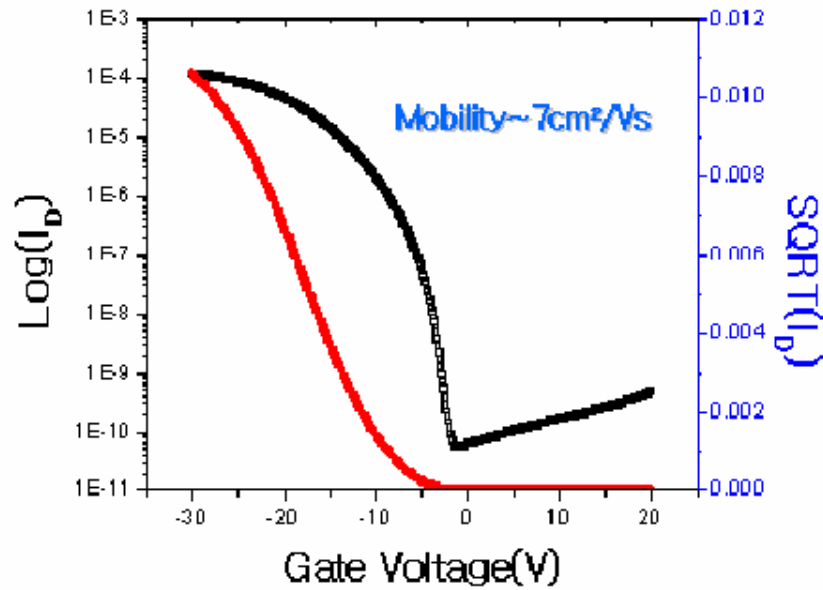
Conductivity vs Mobility in doped amorphous organic semiconductor



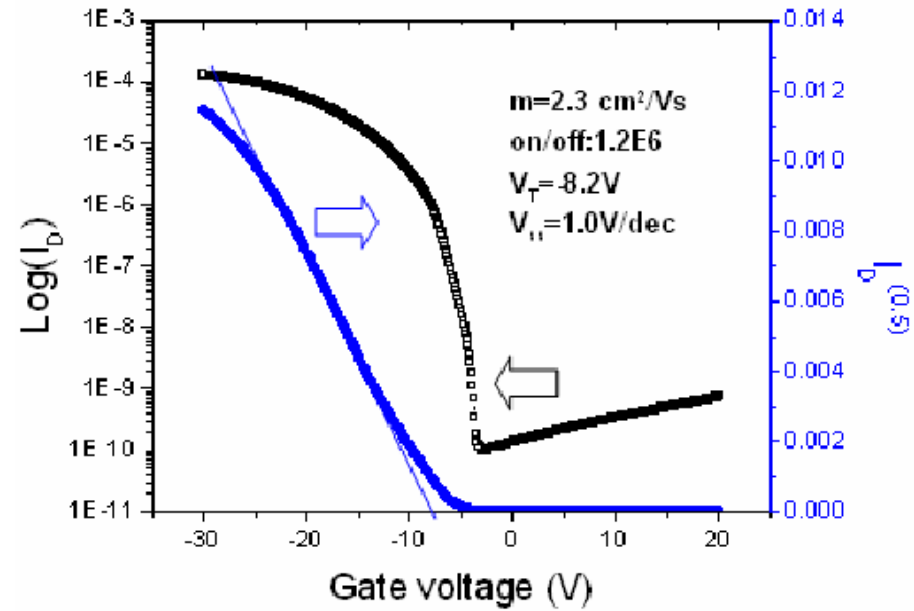
$$\mu \sim \sigma^\delta \quad (\delta < 1)$$

More rapid increase in conductivity with dopant concentration results in decrease of on/off ratio

High performance OTFT array



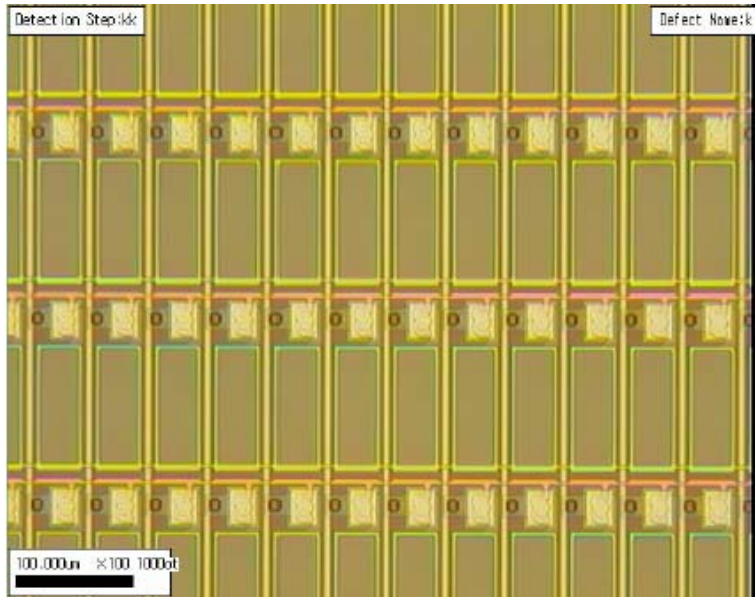
Top contact / Au electrodes
New S4 insulator
(polysilicon-acrylate with titanium complex)



Bottom contact / Au electrodes

High performance OTFT array

Bottom contact / ITO electrodes

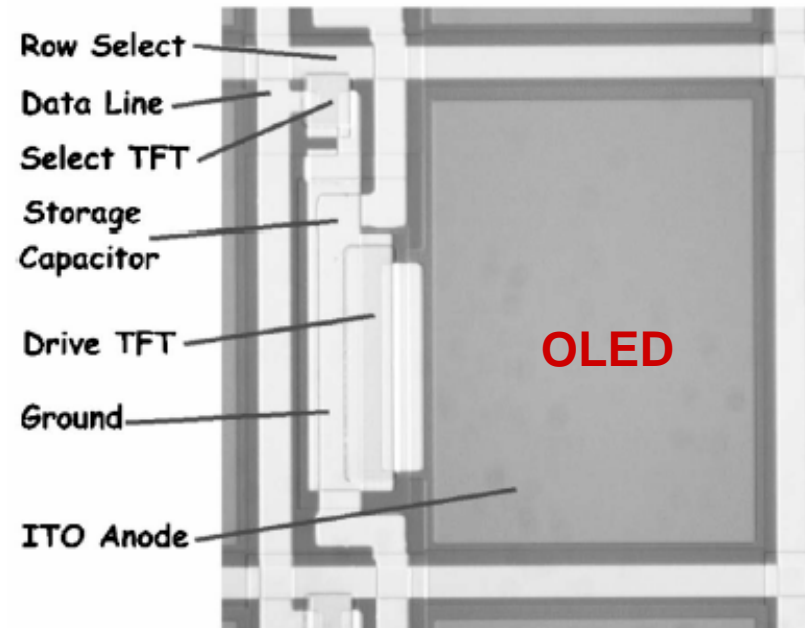
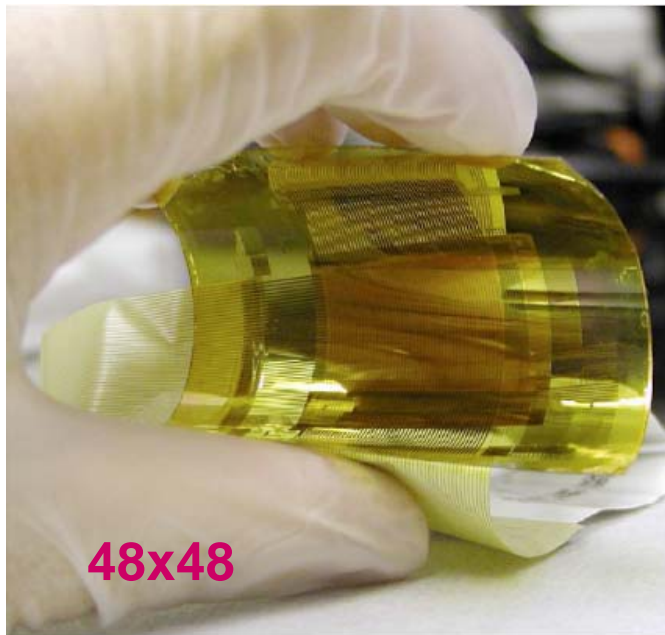


15.0" OTFT array
(Al/Mo gate)



Prototype of 15" XGA OTFT-LCD

All-organic active matrix flexible display

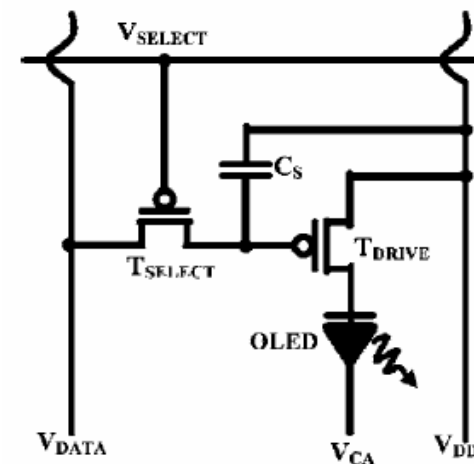


Pixel pitch $500\mu\text{m}$; Aperture ratio:52%
Drive Transistor: $W/L = 200\mu\text{m}/20\mu\text{m}$
Select Transistor: $W/L = 20\mu\text{m}/20\mu\text{m}$

SiN/SiO₂ bi-layer gate (PECVD)
Nickel gate electrode
Pt source and drain electrodes

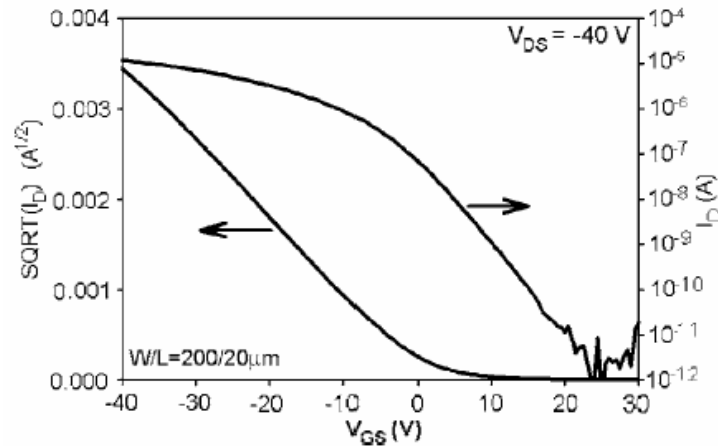
Substrate:

Flexible polyethylene terephthalate (PET)

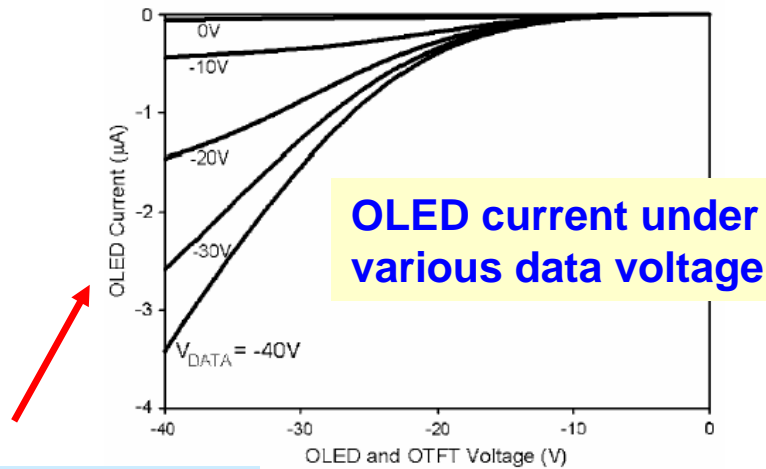


All-organic active matrix flexible display

Before OLED deposition

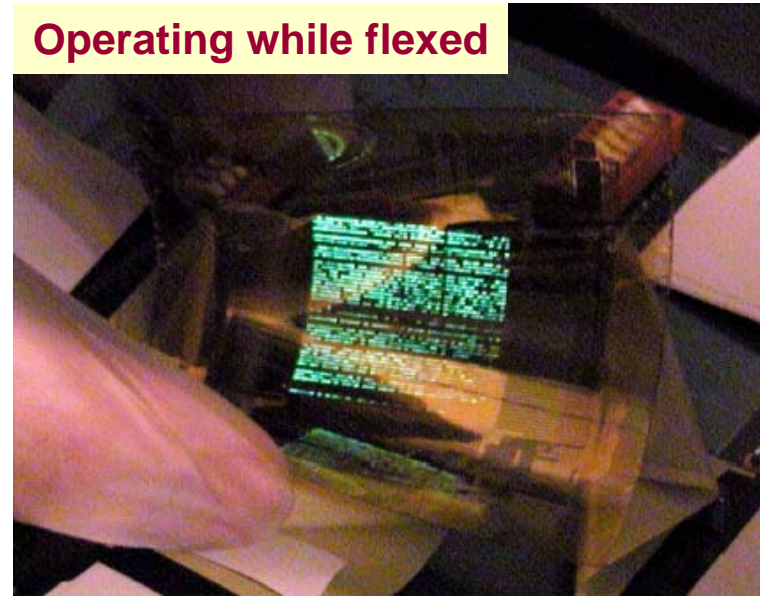


mobility : $0.2 \text{ cm}^2/\text{Vs}$; $V_t = 0.4\text{V}$
on-off ratio : $\sim 10^7$



Good current modulation

Operating while flexed



A pattern of stripes



Fabrication of OTFTs

For most OTFTs,

“Inverted” architecture, in which gate electrode is laid down first

- 1. Electropolymerization**
- 2. Solution-processed deposition**
 - Spin-coating**
 - Ink-jet printing**
- 3. Vacuum evaporation**

Electropolymerization of polythiophene

Starting material, 2,2'-bithiophene

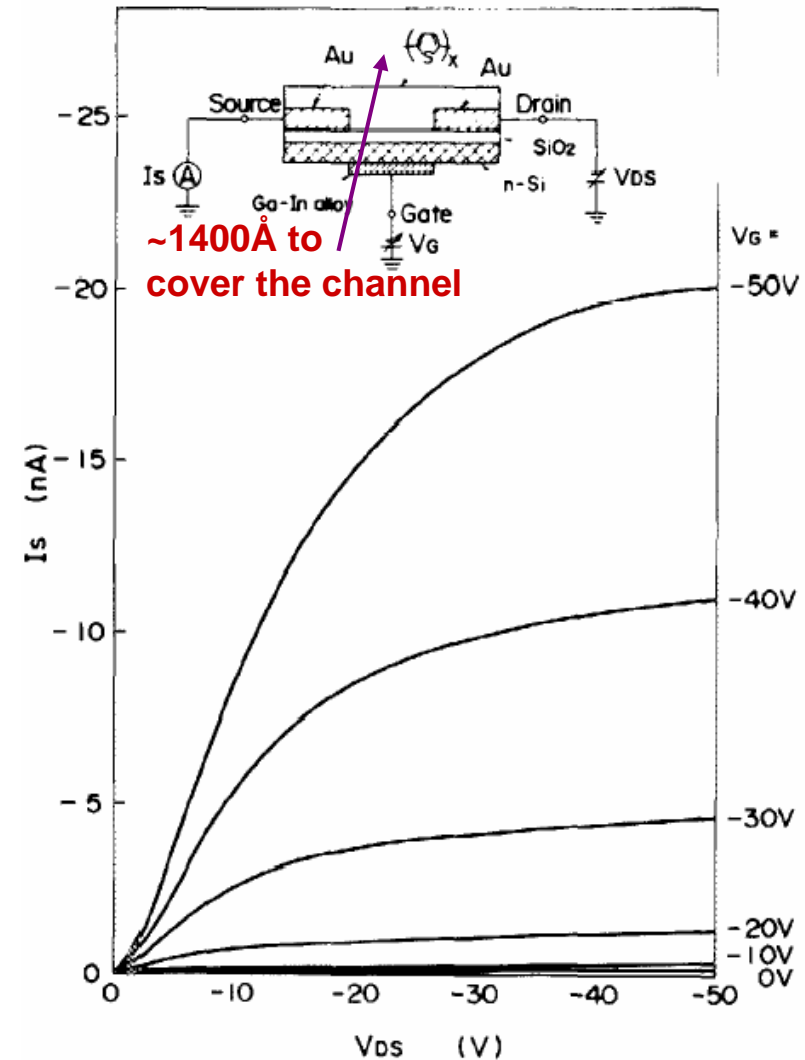
For as-grown polythiophene,
no field-effect was observed

Should be undoped electrochemically firstly

Drawbacks:

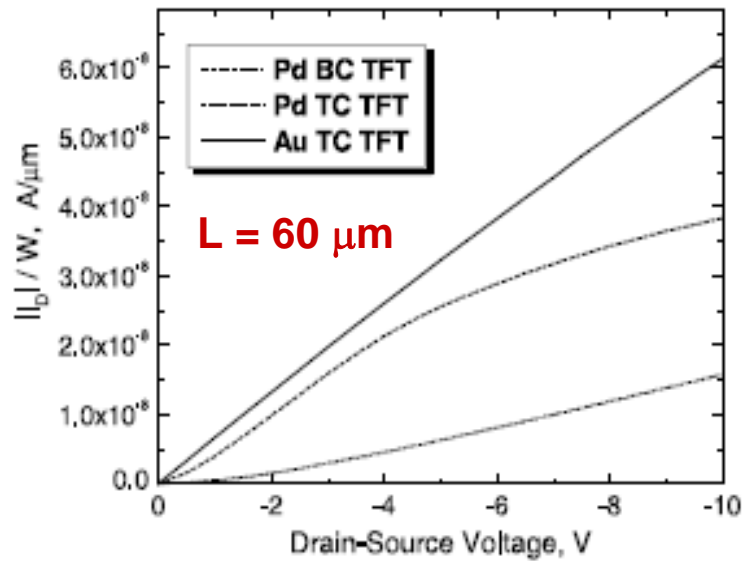
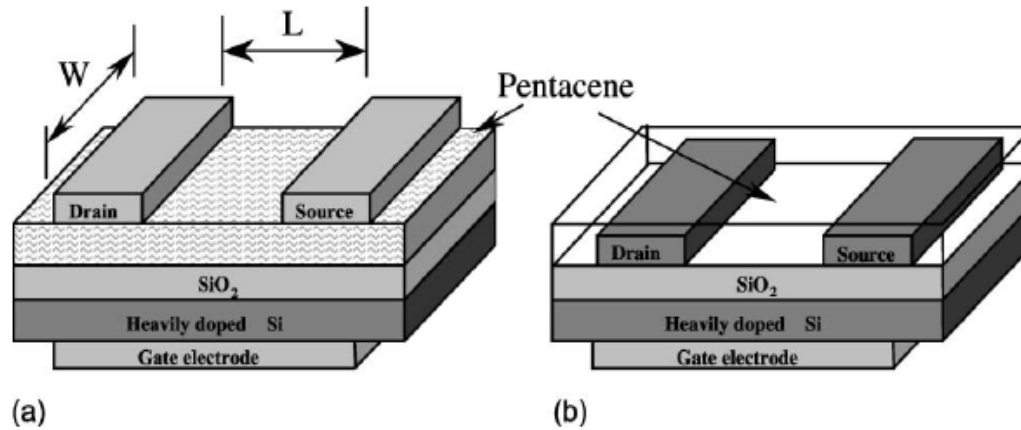
Low mobility, 10^{-5} cm²/V.sec
(high density of structural
and conjugation defects)

Low on-off ratio 10^2 - 10^3

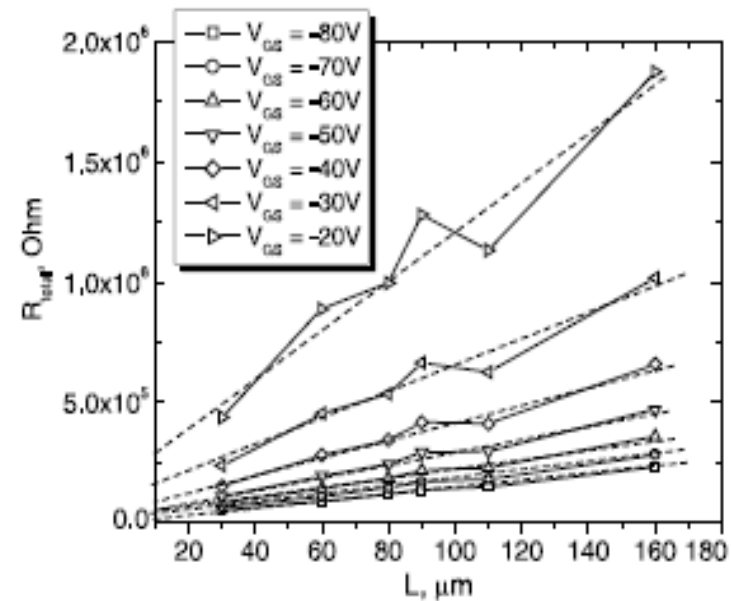


A. Tsumura, and H. Koezuka, and T. Ando, Appl. Phys. Letts., 49, 1210 (1986)

Contact resistance in OTFTs



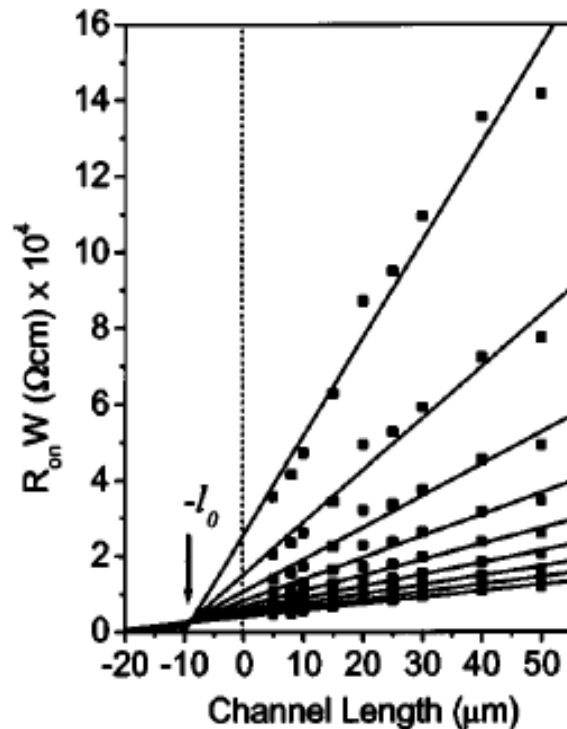
pronounced non-linearity
for Pd TC and BC devices



Transfer line method
Gold TC TFT

Transfer line method

$$R_{on} = \frac{\partial V_{DS}}{\partial I_{DS}} \Big|_{V_G}^{V_{DS} \rightarrow 0} = R_{ch} + R_p = \frac{L}{W\mu_i C_i (V_G - V_{T,i})} + R_p$$

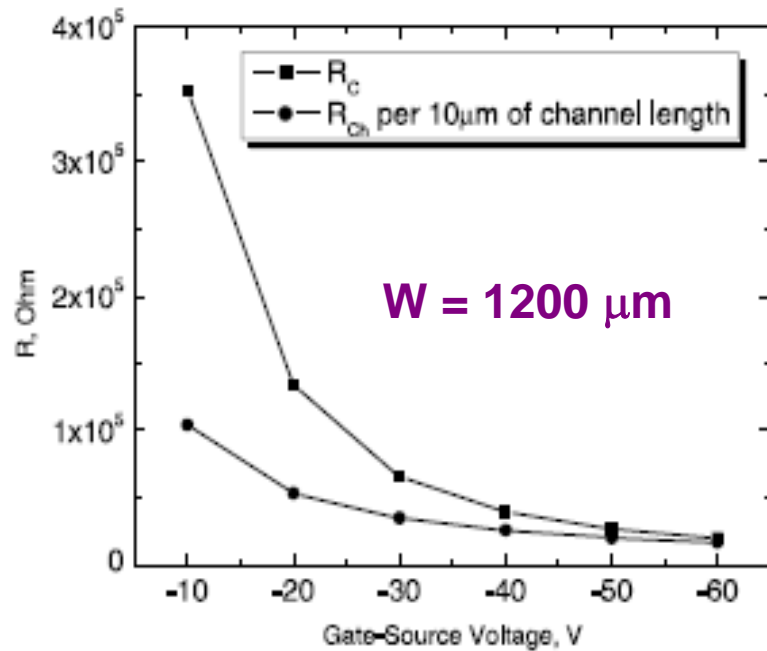


μ_i ,the intrinsic mobility
 $V_{T,i}$,threshold voltage
 R_p ,parasitic resistance
 R_{on} ,device resistance
 R_{ch} ,channel resistance

The contact resistance is extracted by plotting the width-normalized resistance (RW) as a function of L

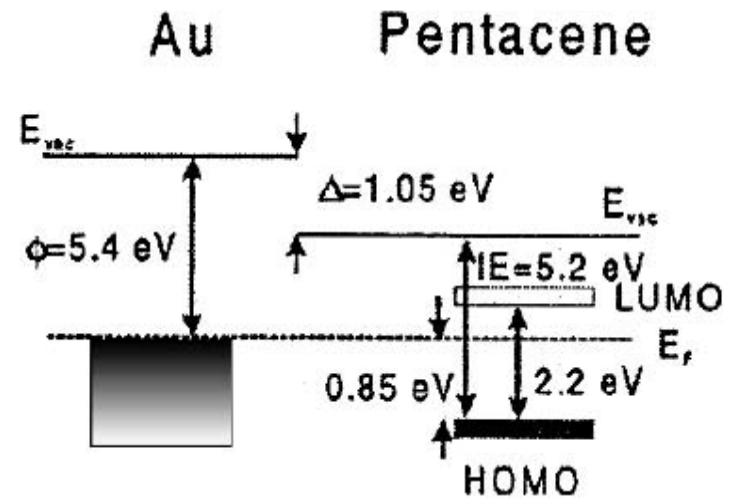
The extrapolation to zero channel length gives the total (S&D) contact resistance

Contact resistance in OTFTs



Channel resistance becomes comparable with the contact resistance at $V_{GS} > 40V$

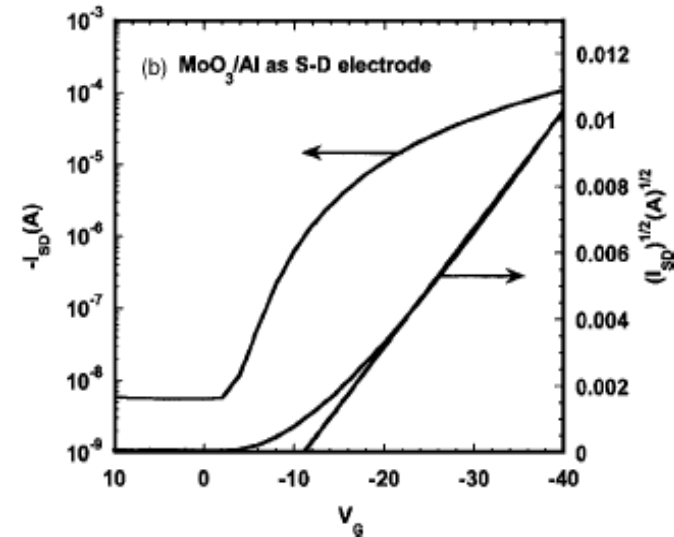
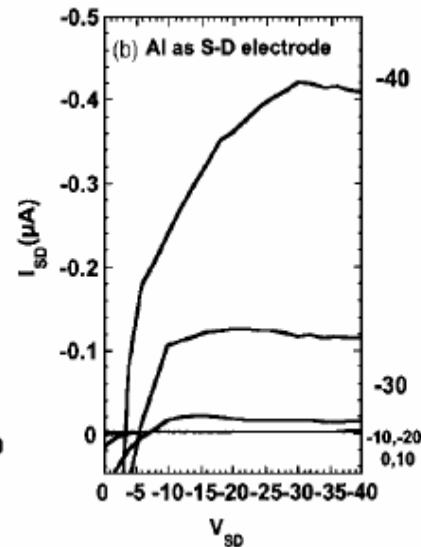
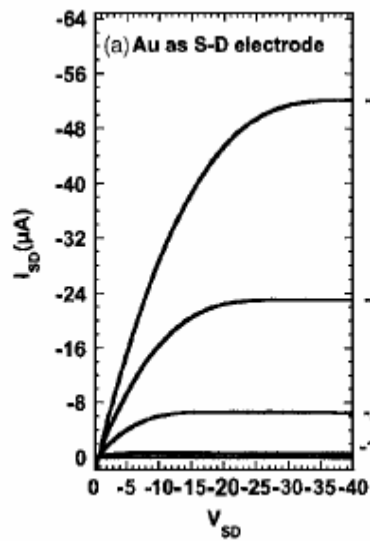
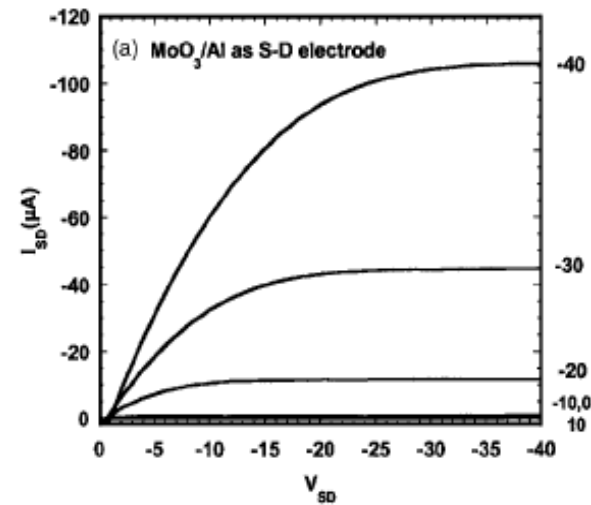
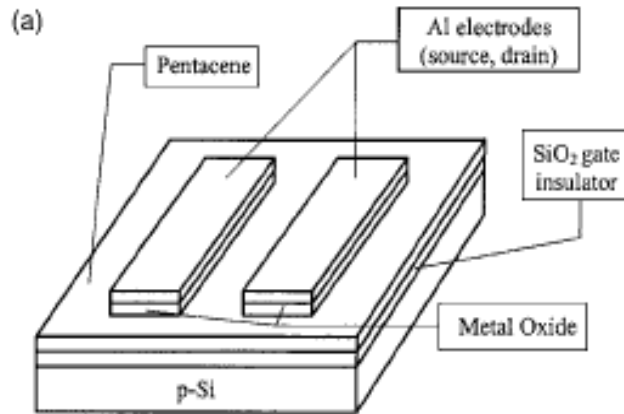
Performance of TFT with $L < 10 \text{ mm}$ can be limited by contacts



Shottky-Mott model is not followed

Adsorbed molecules tend to push back the tailing e^- density at the metal surface, thus reducing the surface dipole and decreasing the work function of the metal

Metal oxide/metal bilayer electrodes for OTFTs



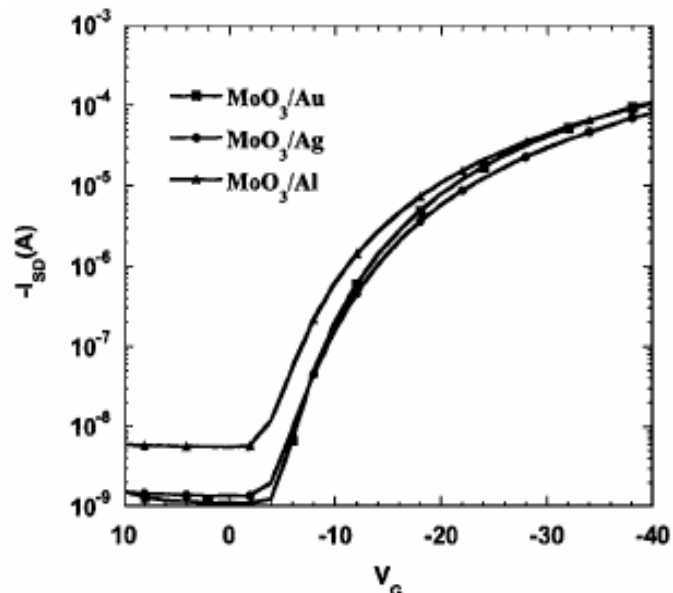
Large injection barrier while Al was used as the electrodes

Improved I-V characteristics after MoO₃ is incorporated

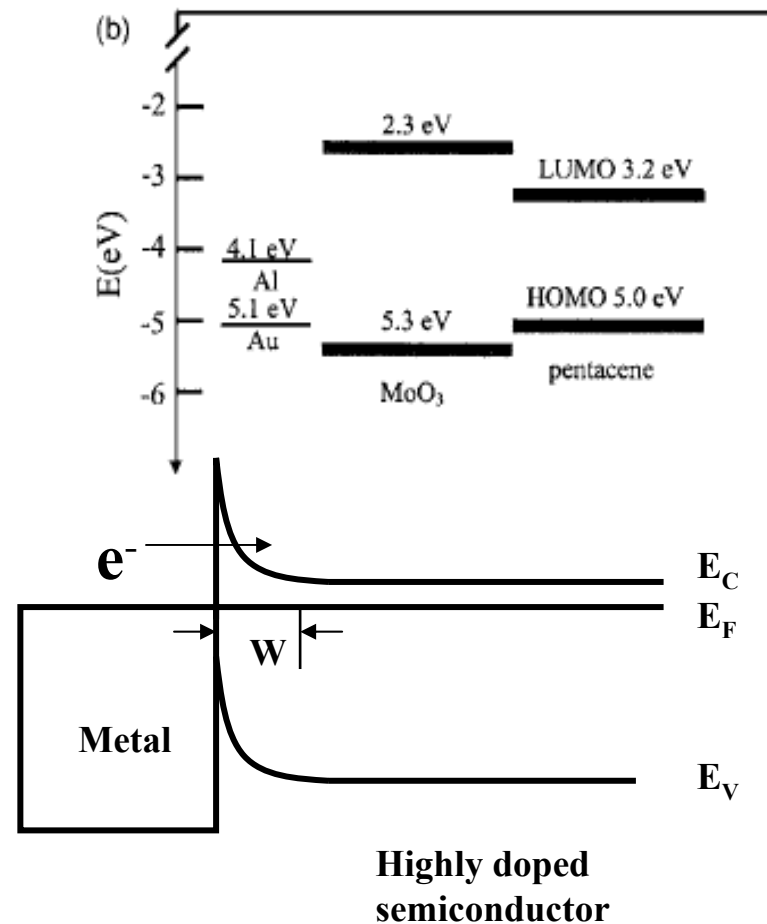
Metal oxide/metal bilayer electrodes for OTFTs

TABLE I. Electrical parameters of the OTFTs in this study.

Drain-source electrodes	Mobility (cm ² /V s)	Threshold voltage (V)	On-off ratio
Al	2.8×10^{-3}	-16.2	2.3×10^2
Au	0.182	-8.75	2.8×10^4
MoO ₃ /Al	0.4	-12.1	3.8×10^4
WO ₃ /Al	0.253	-12.88	4.1×10^4
V ₂ O ₅ /Al	0.226	-10.43	1.8×10^4



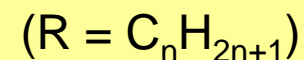
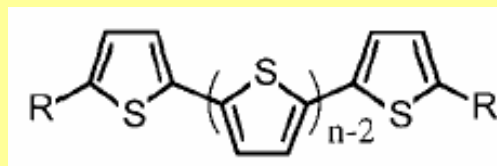
An Ohmic contact is achieved between the metal and MoO₃



MoO₃ yields a thin film containing MoO to MoO₃ as well as Mo, resulting in highly doped semiconductor

Materials for OTFTs

Oligothiophenes



Mobility is “independent” of the present or absent of alkyl end substitution

The first OTFT made with a small conjugated molecule

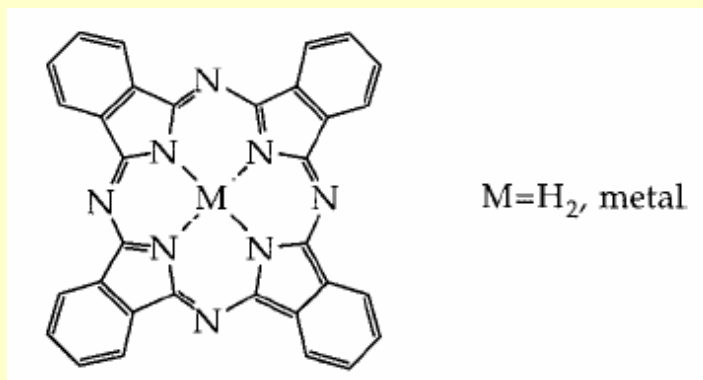
Table 1. Typical mobility [$\text{cm}^2\text{V}^{-1}\text{s}^{-1}$] of recent oligothiophene-based OFETs.

Compound	Unsubstituted	Dihexyl-substituted	Reference
4T	$10^{-4} - 6 \times 10^{-3}$		[35,36]
5T	1.5×10^{-3}		[36]
6T	0.01 - 0.03	0.04 - 0.06	[30-32]
		0.09 - 0.13	[34]
8T	0.01 - 0.03	0.01	[35,37]

Phthalocyanines (PCs)

The first reported organic semiconductors (1948)

Thermally stable up to 400°C and easy to evaporate under vacuum



Mobility : 0.0001 ~ 0.01 cm²V⁻¹s⁻¹

Very sensitive to oxygen

Pentacene

Belong to the family to polyacenes

Table 2. Field-effect mobility [$\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$] of pentacene with various modes of deposition and substrate temperature (RT: room temperature).

Deposition mode	Substrate temperature [$^{\circ}\text{C}$]	Mobility	Reference
Vacuum evaporation	RT	0.002	[48]
Spin-coated precursor		0.001	[51]
Pulsed-laser	RT	0.03	[52]
Vacuum evaporation	RT	0.038	[49]
Vacuum evaporation	85	0.4	[53]
Vacuum evaporation	120	0.62	[50]

On-off ratio : up to 10^8

properties:

- *sublime at $290\text{ }^{\circ}\text{C}$ - $300\text{ }^{\circ}\text{C}$
- *decompose $> 300\text{ }^{\circ}\text{C}$ in air
- *do not dissolve in water
- *slightly dissolved in organic solvents
- *optical band gap : 2.8 eV



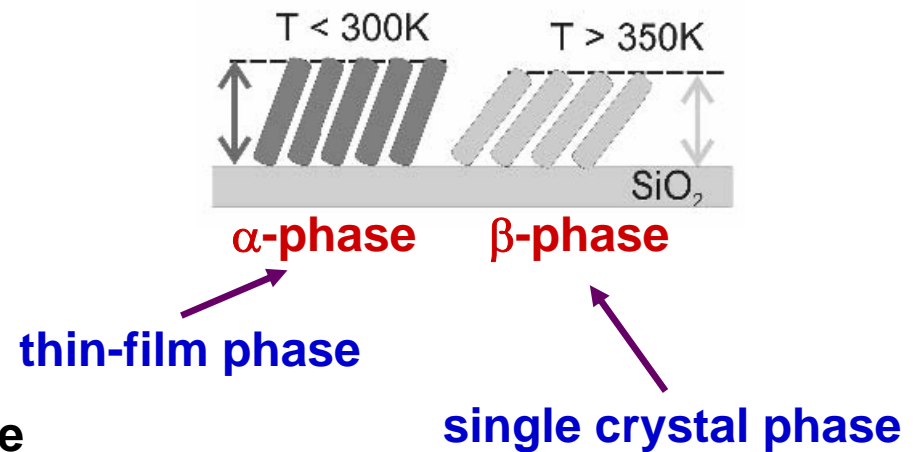
Pentacene

In bulk, triclinic phase
(β -phase, $d_{001} = 14.5\text{\AA}$)

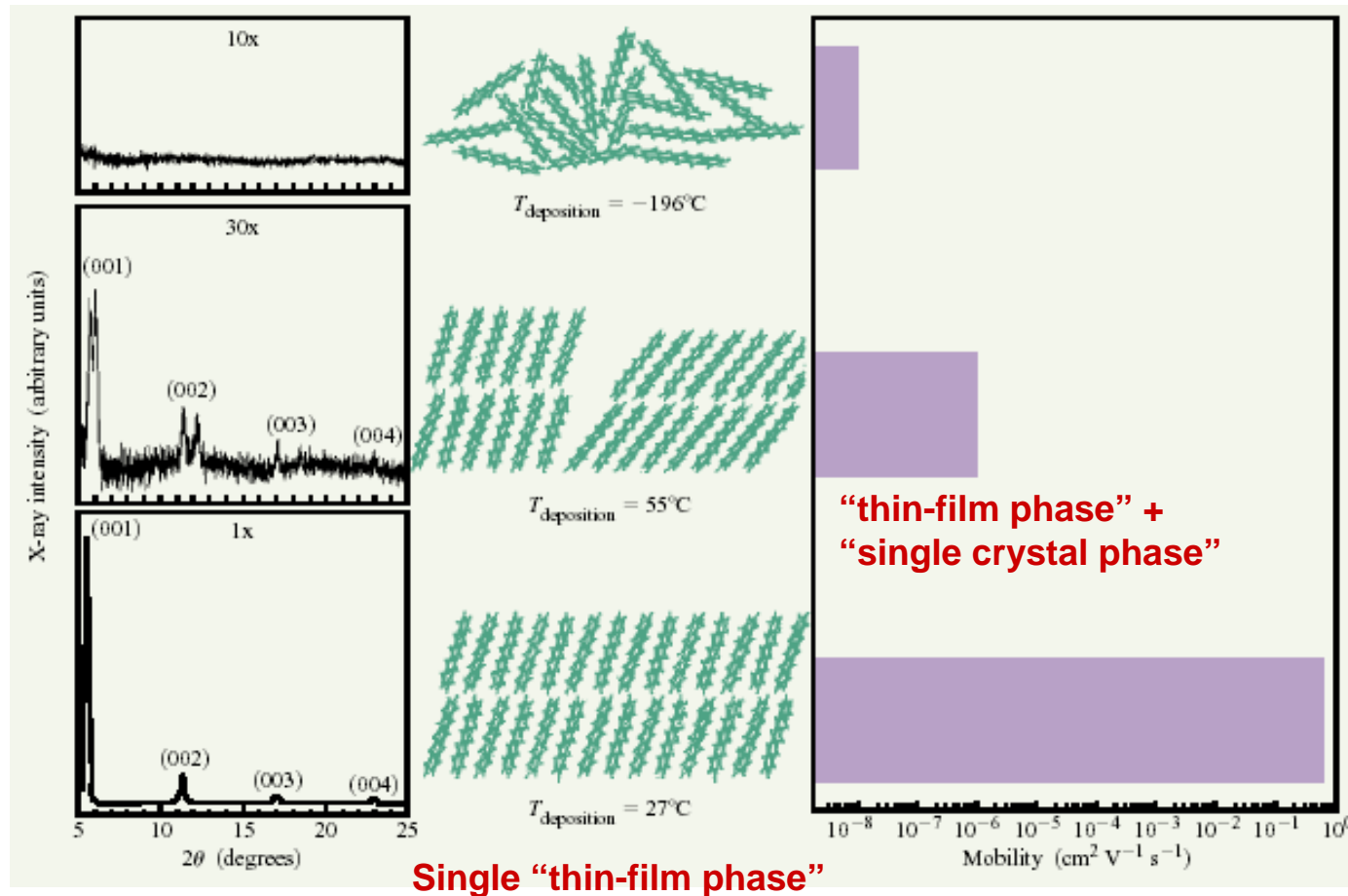
On SiO_2 surface
metastable, surface induced structure
(α -phase, $d_{001} = 15.5\text{\AA}$)
called “thin-film structure”
(due to the low (001) surface energy)

Length of pentacene = 16\AA

Different of in molecular packing will affect dramatically
the electronic properties



Pentacene



X-ray diffractograms, and schematic representations of structural order of pentacene at different substrate temperature

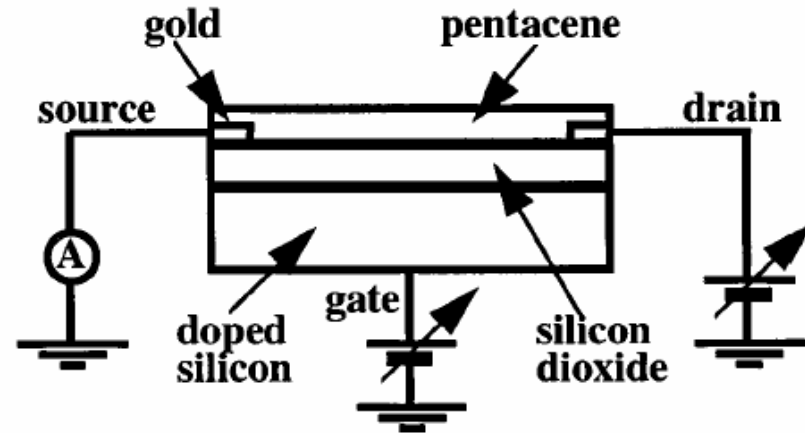
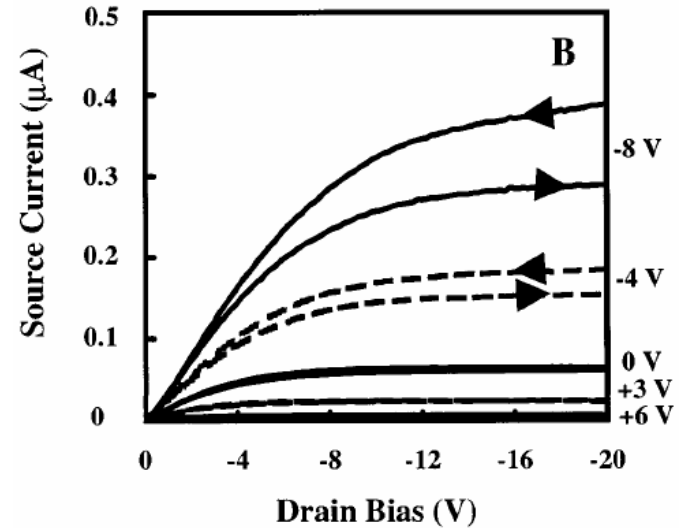
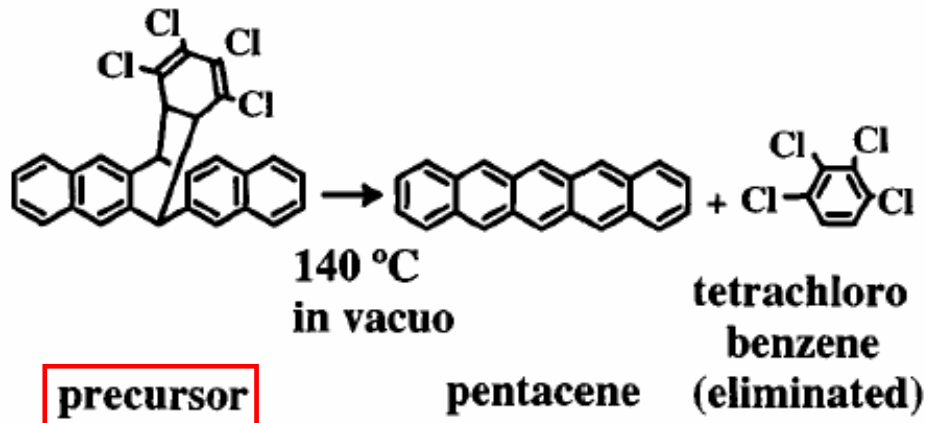
Solution Processed Pentacene

Mobility : $9 \times 10^{-3} \text{ cm}^2/\text{Vs}$

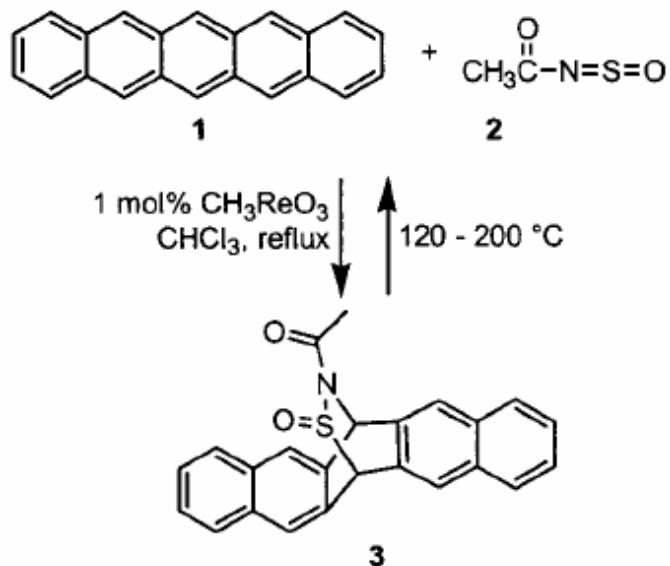
On-off ratio $\sim 10^5$

Advantages :

low temperature process
can be patterned by IJP



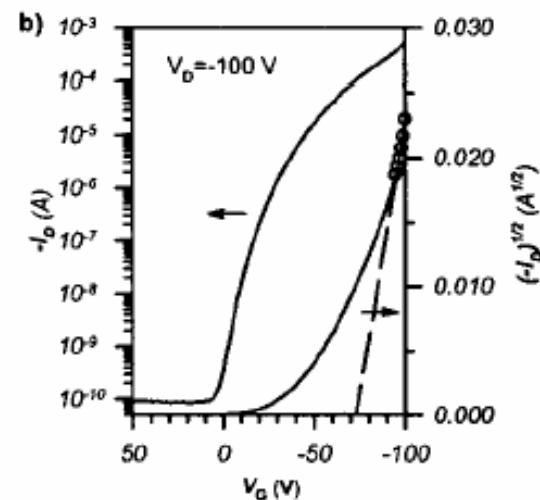
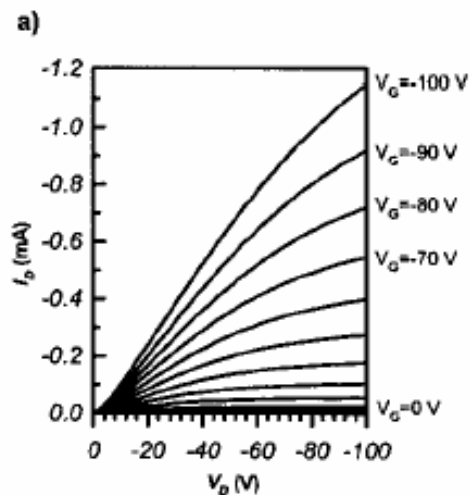
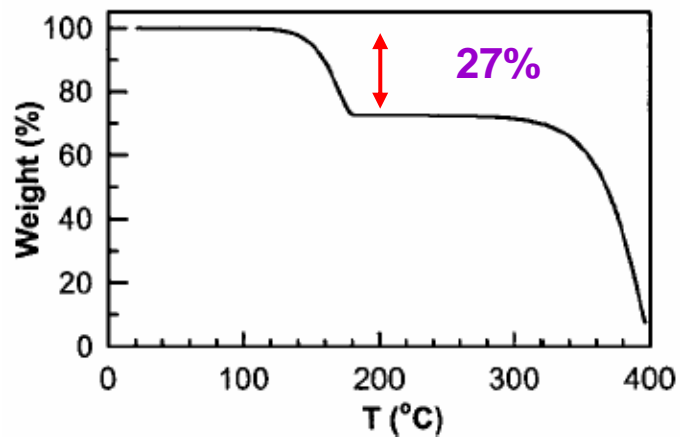
Solution Processed Pentacene



Single step synthesis

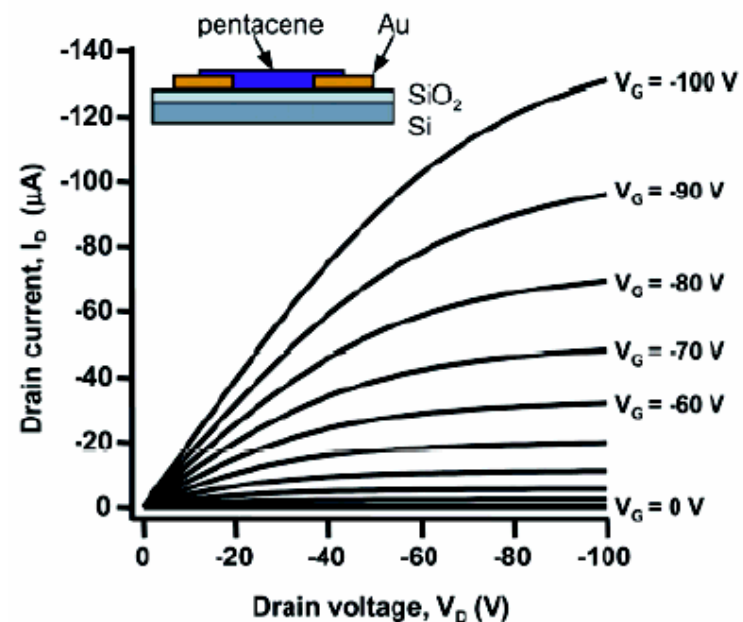
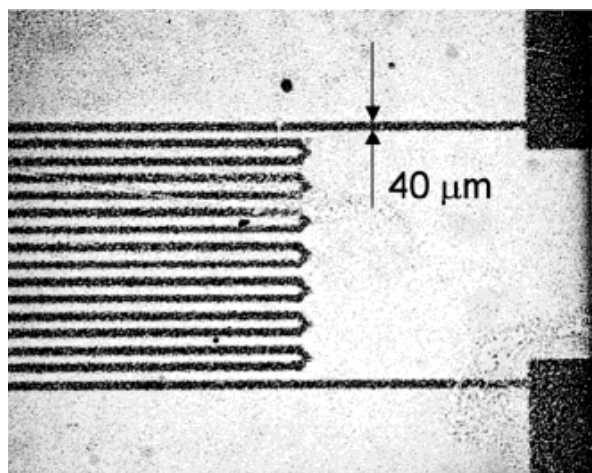
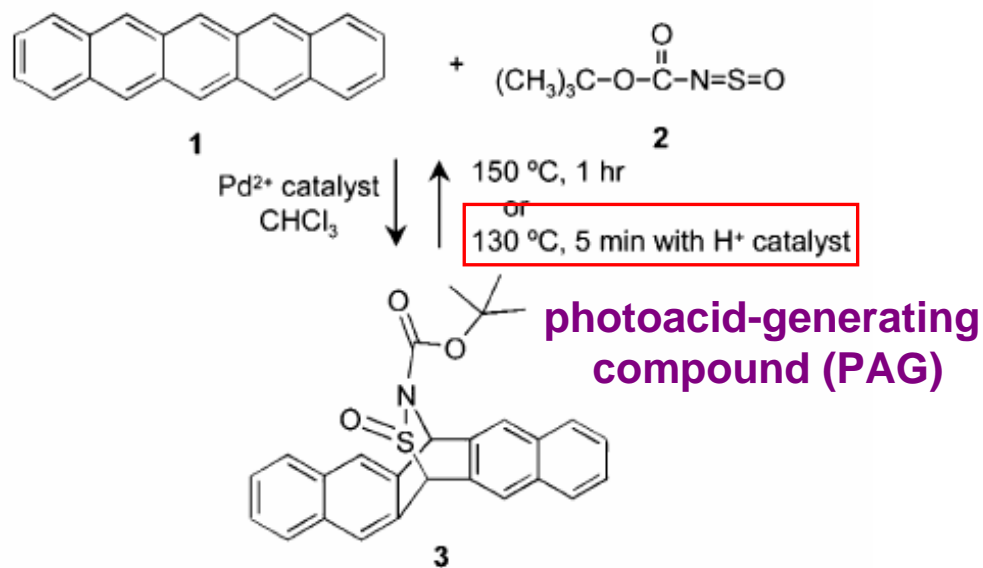
Mobility : 0.29 cm²/Vs (linear region)
0.42 cm²/Vs (saturation region)

On-off ratio : 2x10⁷



Bottom contact, 200°C 1.5 min under N₂

Photopatternable Pentacene



Mobility : 0.13 cm²/Vs (linear region)
0.25 cm²/Vs (saturation region)
On-off ratio : 2x10⁵

Pentacene patterned by UV exposure through a mask and heating in the presence of PAG

N-type semiconductors

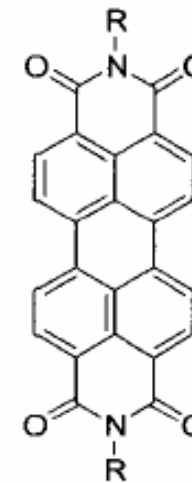
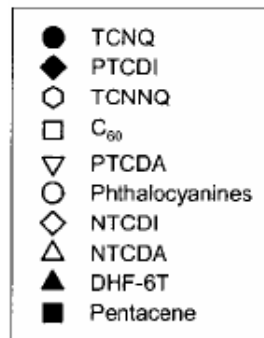
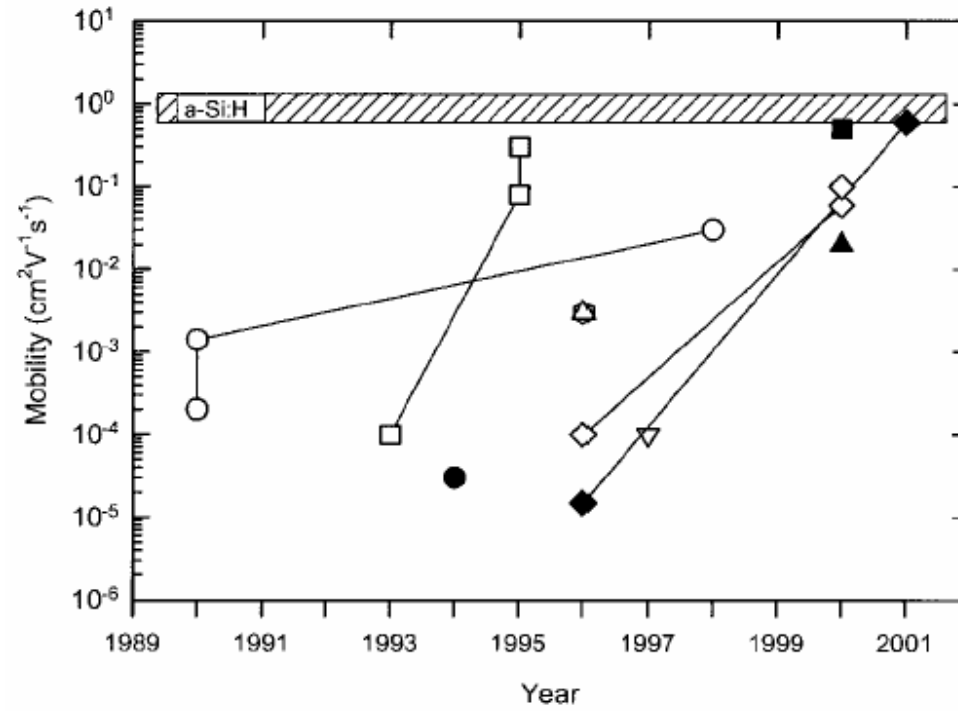
OLED research area introduce the concept of electron and hole transporting materials.

n- and p- type materials are usually characterized by their high electron affinity and low ionization potential, respectively.

In OTFT, the *accumulation* region is set up for **positive** gate biases for n- type materials, and for **negative** gate biases for p-type materials

**Problem : strong instability with respect to oxygen and water
(act as electron traps)**

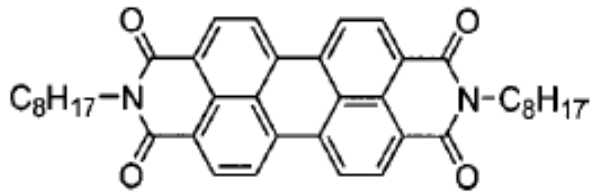
N-type semiconductors



PTCDI-C8

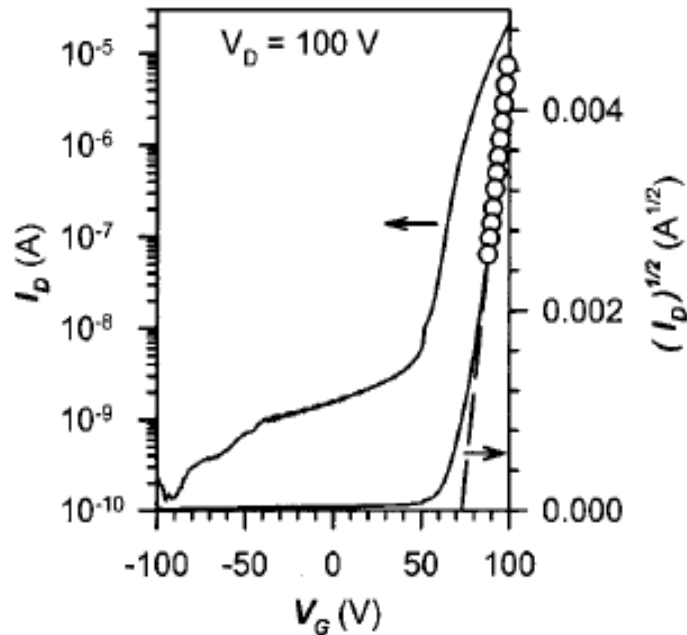
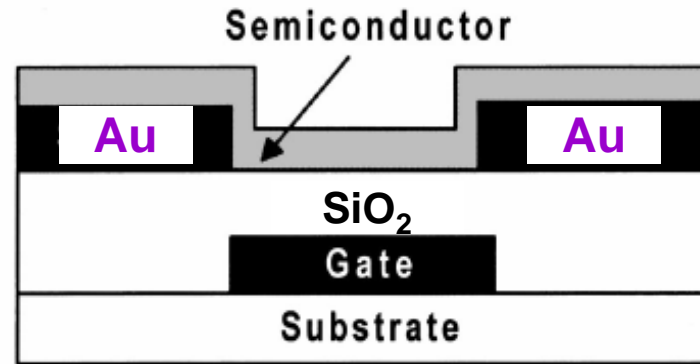
mobility : 0.6 cm²/Vs
on-off ratio : ~ 10⁵

High mobility n-type OTFTs

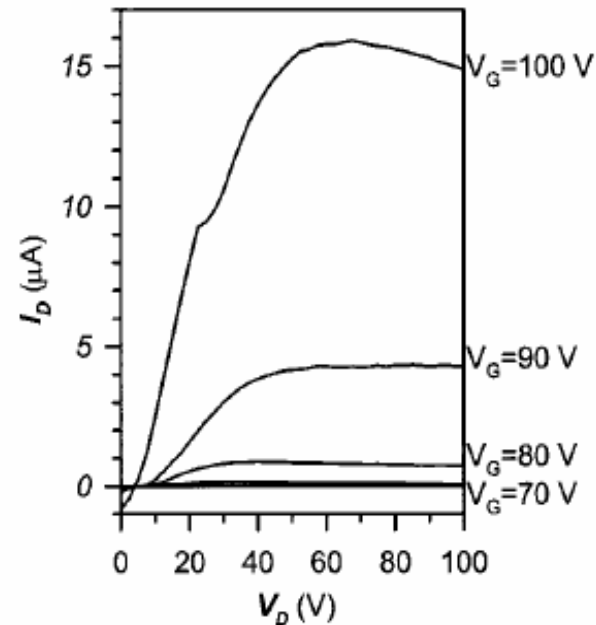


PTCDI-C8H

Smectic liquid crystalline phases

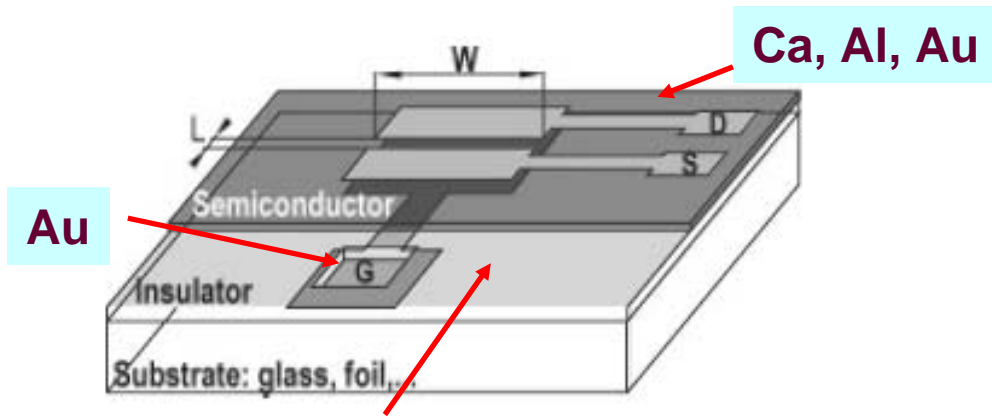


Mobility : 0.6 cm²/Vs
 $V_t = 75V$
 On-off ratio > 10⁵

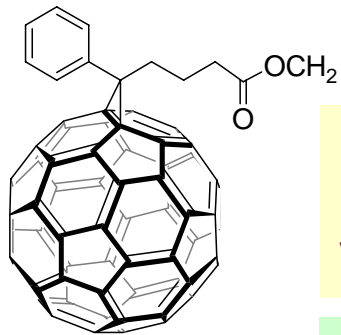


Mobility : 0.3 cm²/Vs in the linear region

Solution-processed n-type OTFTs



Organic resin



PCBM
n-type

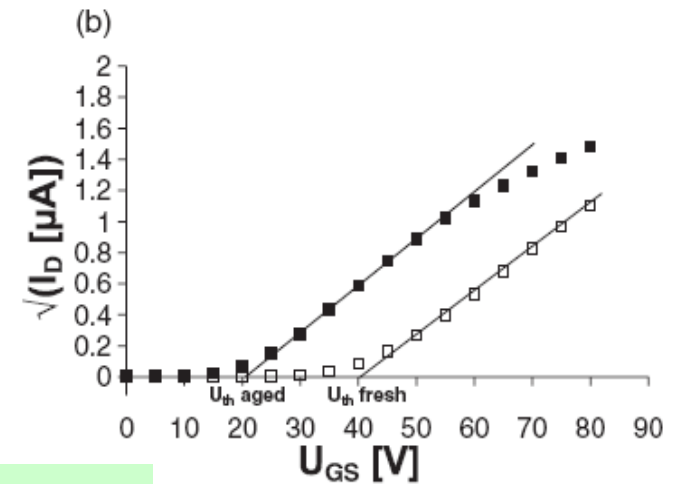
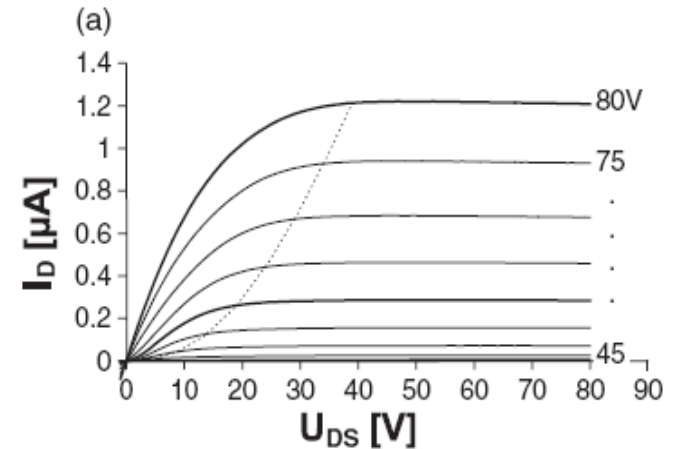
Mobility : $3.5 \times 10^{-3} \text{ cm}^2/\text{Vs}$
(saturation region)

$V_t = 41\text{V}$

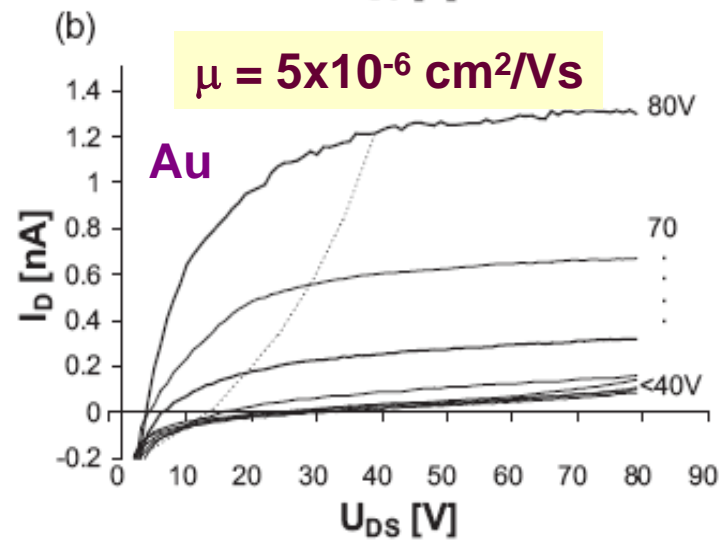
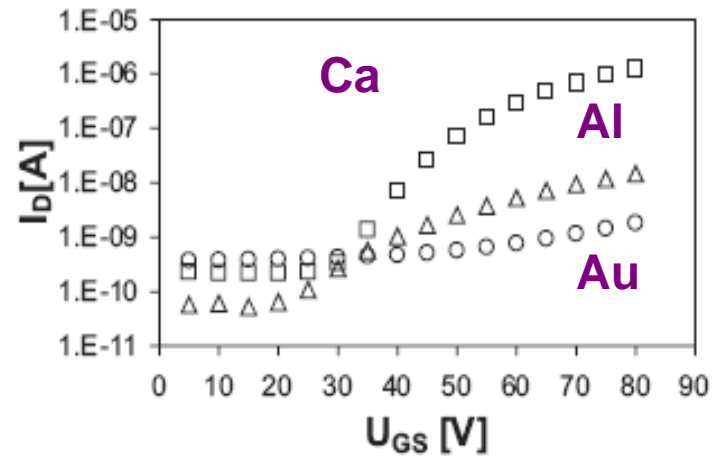
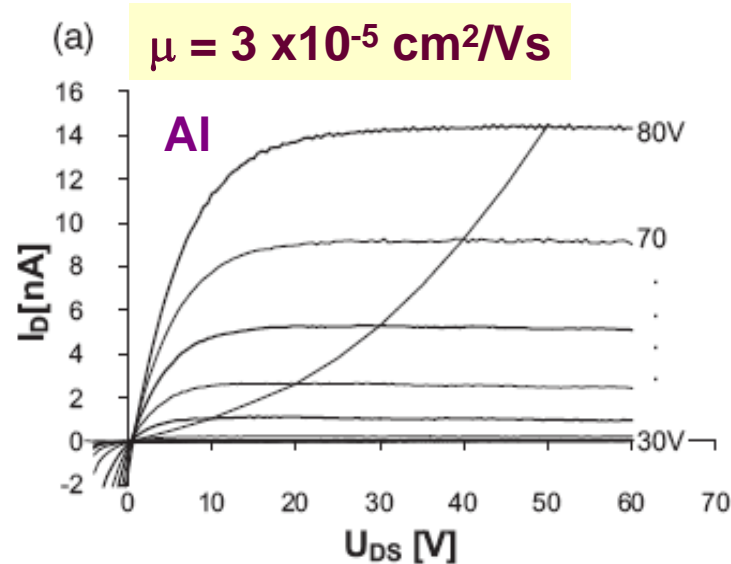
After one work storage
Mobility : $4 \times 10^{-3} \text{ cm}^2/\text{Vs}$
(saturation region)

$V_t = 21\text{V}$

May be due to trap filling by reduction



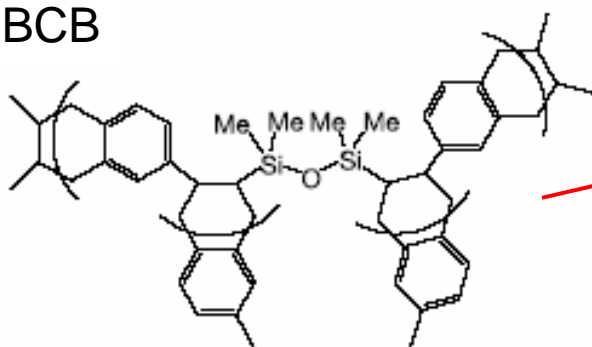
Solution-processed n-type OTFTs



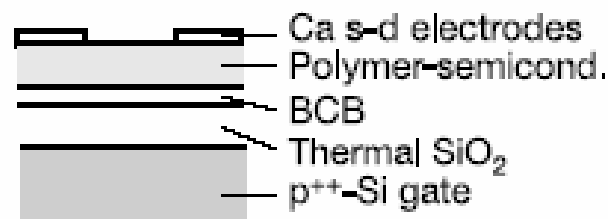
Threshold voltage seems unaffected
Fermi-level pinning (due to traps)
A higher work function of the contact materials leads decreased injection

General observation of n-type behavior of organic semiconductors

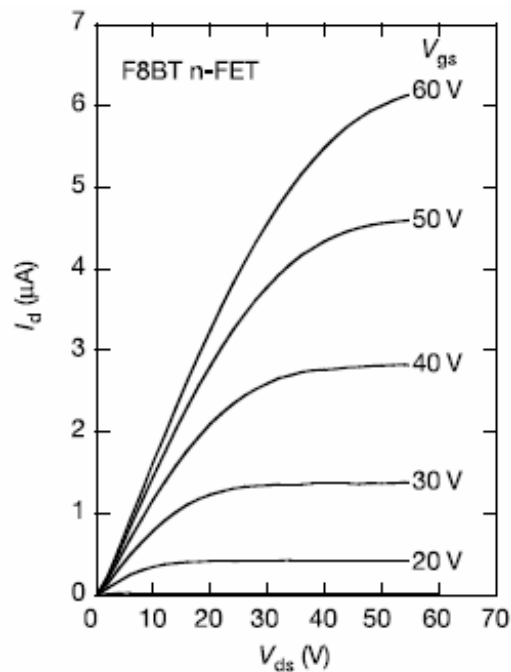
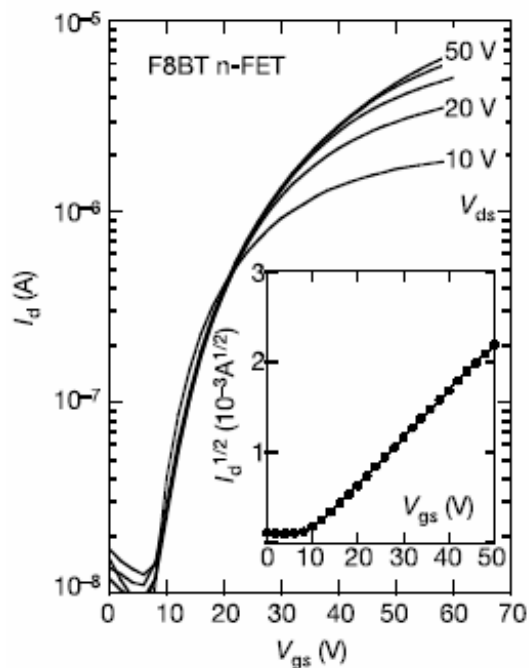
BCB



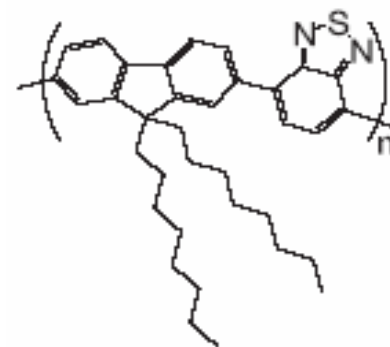
Device Structure



Low-work function (Ca) was used to enable determination of true mobility without correcting for the contact effect



F8BT Poly(9,9-dioctylfluorene-*alt*-benzothiadiazole)



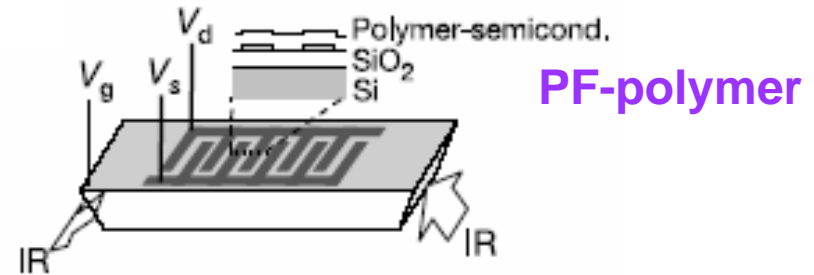
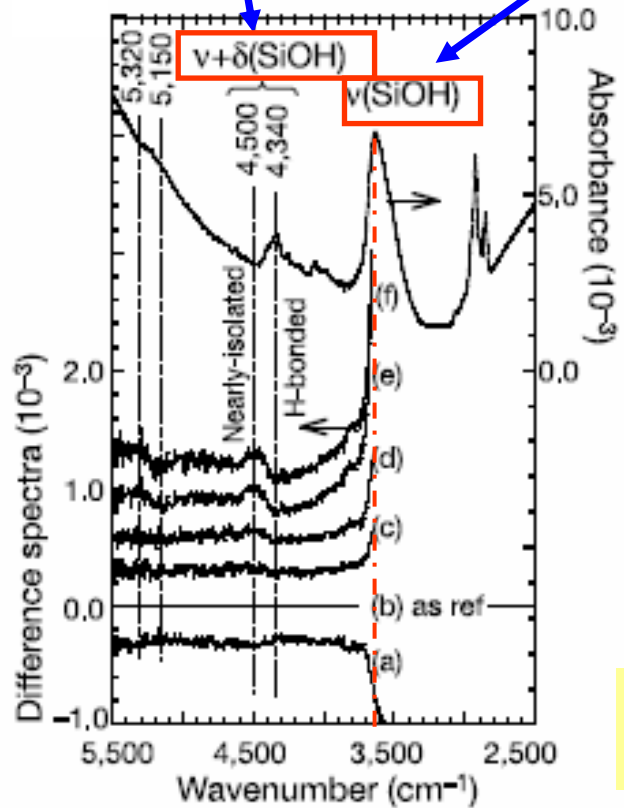
$$\mu_e = 4 \times 10^{-3} \text{ (130}^\circ\text{C)}$$

$$= 5 \times 10^{-3} \text{ (220}^\circ\text{C)}$$

Interfacial electron trapping on standard Si/SiO₂ backgate device

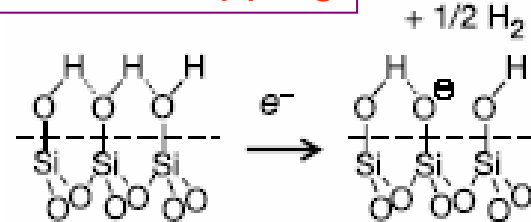
Stretching + Bending

stretching

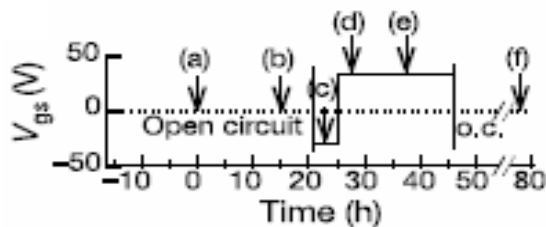


multiple-reflection attenuated-total-reflection FTIR spectrometry

Mechanism of e⁻ trapping

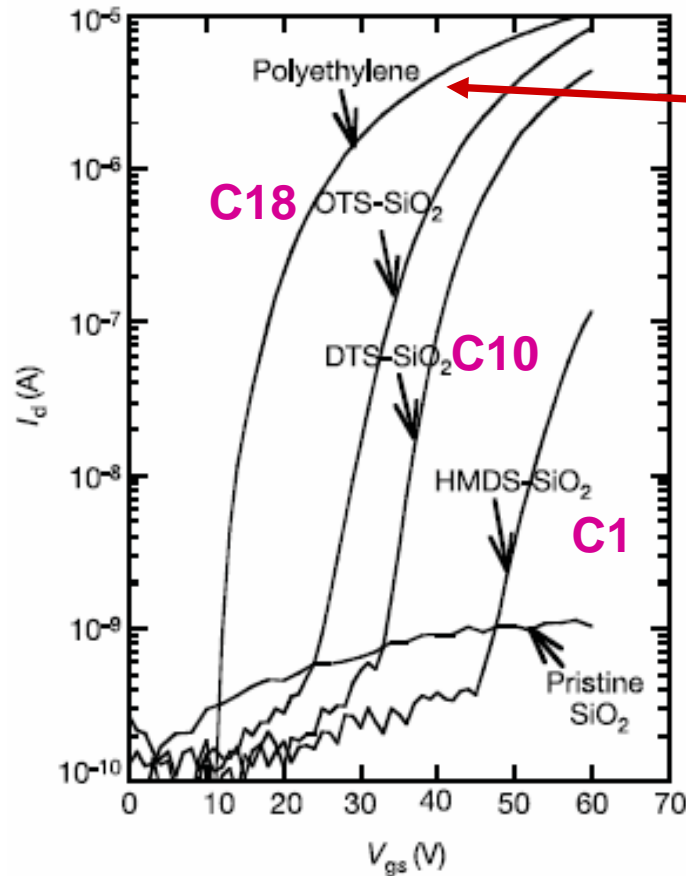
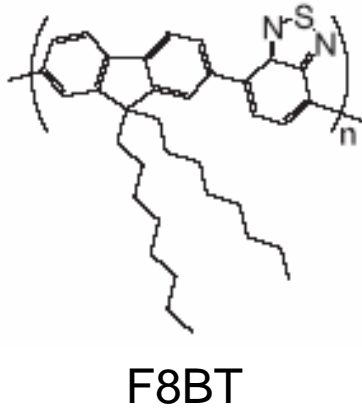


Typical SiOH conc. = $\sim(3-7) \times 10^{13} \text{ cm}^{-2}$ (6~12% monolayer)
 Typical carrier conc. in normal FET operation $\sim 10^{13} \text{ cm}^{-2}$



Si-OH intensity shift from H-bonded (4,340cm⁻¹) to nearly-isolated (4,500cm⁻¹) as soon as a positive bias was applied → e⁻ trapping

F8BT n-channel FETs – further evidence



Stable n-channel conduction was observed, as is the case with BCB

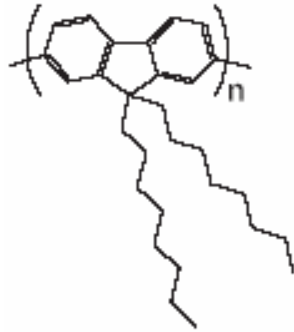
With SAM passivation, n-channel activity can be observed

In the past, SiO₂ or hydroxy-containing polymers, such as poly(vinyl phenol) and polyimides (-COOH), were used as the gate dielectrics, the severe e- trapping suppress n-channel conduction

Summary

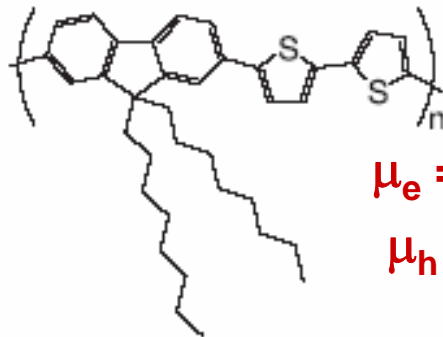
Gate : BCB; electrodes: Ca
(n-channel)

F8 Poly(9,9-dioctylfluorene)



$$\begin{aligned} \mu_e &= 5 \times 10^{-3} \text{ (130}^\circ\text{C)} \\ &= 1 \times 10^{-2} \text{ (240}^\circ\text{C)} \\ \mu_h &= 3 \times 10^{-4} \text{ (200}^\circ\text{C)} \\ &\text{(HMDS-SiO}_2\text{, Au)} \end{aligned}$$

F8T2 Poly(9,9-dioctylfluorene-*alt*-bithiophene)



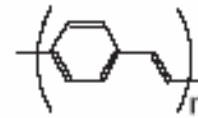
$$\begin{aligned} \mu_e &= 6 \times 10^{-3} \text{ (130}^\circ\text{C)} \\ \mu_h &= 5 \times 10^{-3} \text{ (200}^\circ\text{C)} \\ &\text{(HMDS-SiO}_2\text{, Au)} \end{aligned}$$

P3HT Regioregular poly(3-hexylthiophene)



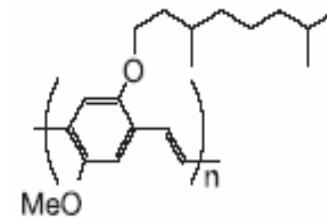
$$\begin{aligned} \mu_e &= 6 \times 10^{-4} \text{ (100}^\circ\text{C)} \\ \mu_h &= 2 \times 10^{-4} \text{ (200}^\circ\text{C)} \\ &\text{(BCB, Ca)} \end{aligned}$$

PPV Poly(*p*-phenylenevinylene)



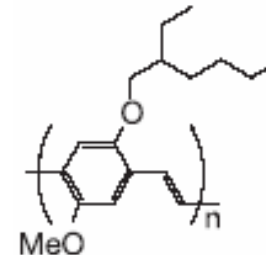
$$\mu_e = 1 \times 10^{-4} \text{ (250}^\circ\text{C)}$$

OC₁₀-PPV Poly(2-methoxy-5-(3,7-dimethyloctyloxy)-*p*-phenylene vinylene)



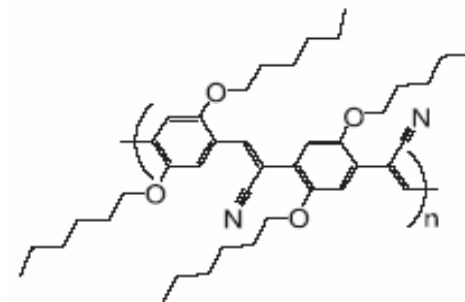
$$\begin{aligned} \mu_e &= 8 \times 10^{-5} \text{ (130}^\circ\text{C)} \\ &= 2 \times 10^{-3} \text{ (200}^\circ\text{C)} \\ \mu_h &= 5 \times 10^{-4} \text{ (200}^\circ\text{C)} \\ &\text{(BCB, Au)} \end{aligned}$$

MEH-PPV Poly(2-methoxy-5-(2-ethylhexyloxy)-1,4-phenylene vinylene)



$$\begin{aligned} \mu_e &= 3 \times 10^{-5} \text{ (200}^\circ\text{C)} \\ \mu_h &= 5 \times 10^{-5} \text{ (200}^\circ\text{C)} \\ &\text{(BCB, Au)} \end{aligned}$$

CN-PPV Poly(2,5-dihexoxy- α,α' -dicyano-*p*-xylylidene-*alt*-2,5-dihexoxy-*p*-xylylidene)



$$\mu_e = 4 \times 10^{-5} \text{ (130}^\circ\text{C)}$$

Polymers

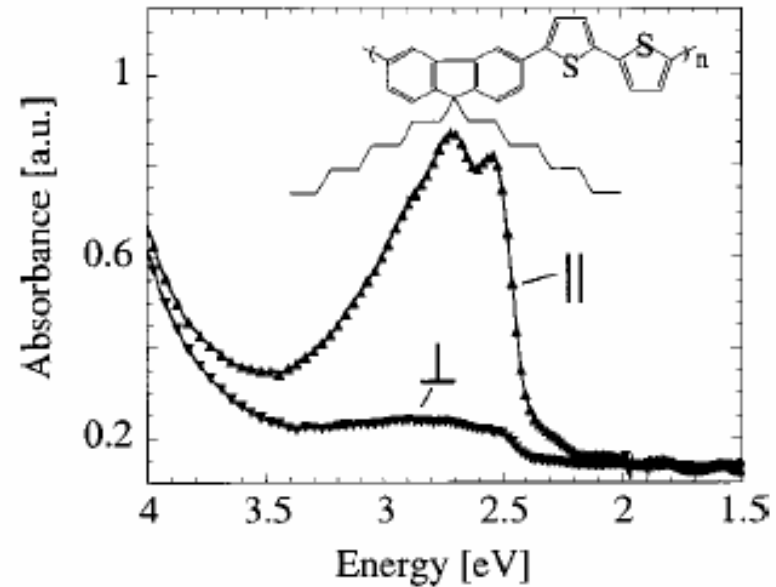
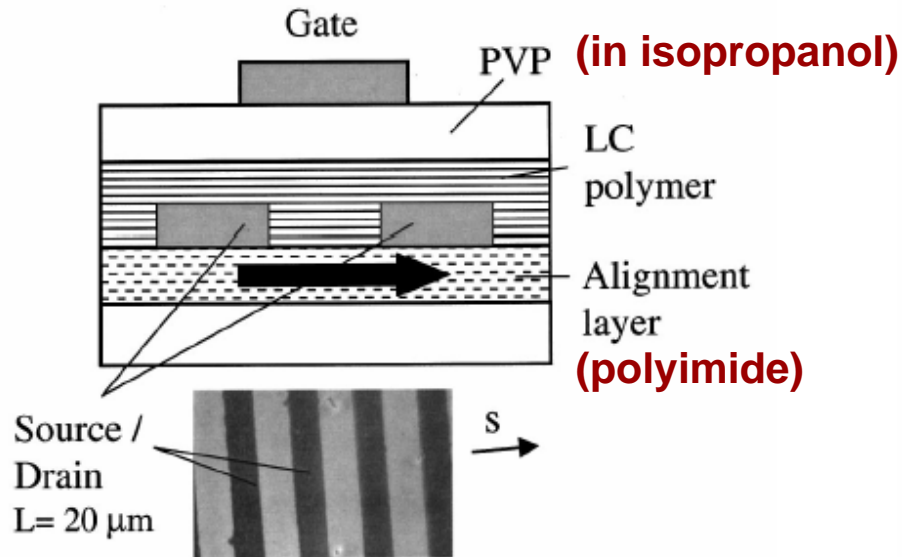
The performance is usually one-order lower than small molecules.

The mobility can be increased by doping, however, at the expense of the on-off ratio

Mobility-conductivity relationship, $\mu \propto \sigma^\delta$ ($\delta \sim 0.7$)

Two major materials
polyfluorene and poly(alkyl-thiophene)

Poly(fluorene)



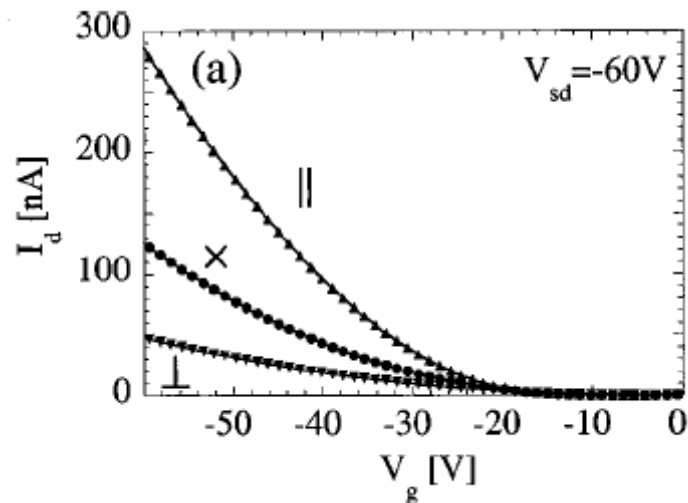
poly(9,9-dioctyl-fluorene-co-bithiophene) (F8T2)

liquid-crystalline self-organization
in rigid-rod nematic conjugated polymers
($T_m = 265^\circ\text{C}$)

$\mu_{||} = 0.009\text{-}0.02 \text{ cm}^2/\text{Vs}$ (linear region)

$\mu_{||} > \mu_x > \mu_{\perp}$

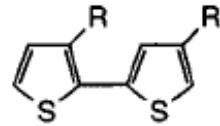
On-off ratio : $> 10^5$



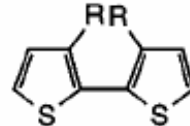
Poly(alkyl-thiophene)

Amorphous films are obtained by spin-coating for regiorandom polymers (both HT and HH)

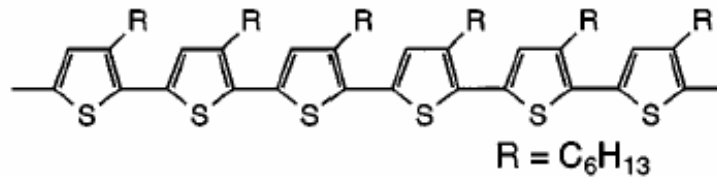
regioregularity



head-to-tail (HT)



head-to-head (HH)

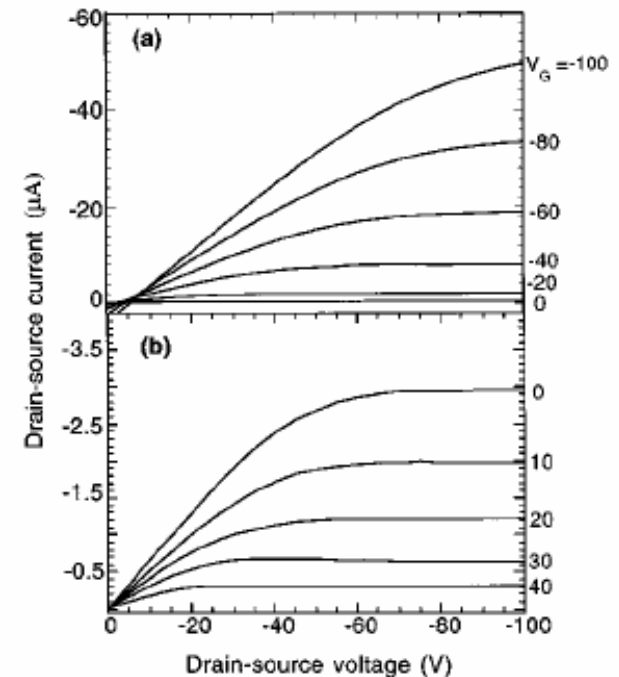
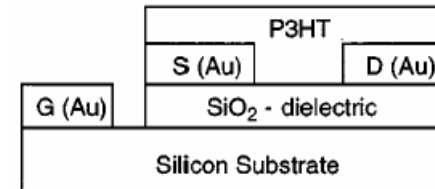


regioregular poly(3-hexylthiophene) (P3HT)
R = C₆H₁₃

Regioregular polymer can be obtained via structure-controlled syntheses

→ Polycrystalline films

mobility : 0.045 cm²/Vs in the accumulation mode
0.01 cm²/Vs in the depletion
on-off ratio : > 10³



Poly(alkyl-thiophene)

TABLE I. Field-effect mobilities and on/off ratios of samples prepared from different conditions. Condition 1, cast, vacuum pumped for 24 h; condition 2, spin-coated; condition 3, treated with NH₃ for 10 h; condition 4, heated to 100 °C under N₂ for 5 min; condition 5, heated to 150 °C under N₂ for 35 min.

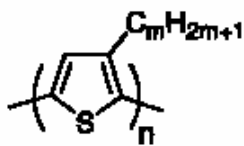
Entry	Solvent	Condition	Mobility (cm ² /V s) ^a	On/off ratio ^b
1	THF	1	6.2×10 ⁻⁴	10
2	<i>p</i> -xylene	1	1.9×10 ⁻³	40
3		2	1.9×10 ⁻⁵	2
4	Toluene	1	3.6×10 ⁻³	10
5		2	3.2×10 ⁻³	25
6	Chlorobenzene	1	4.7×10 ⁻³	10
7		entry 6 condition 3	4.7×10 ⁻³	80
8		2	6.9×10 ⁻⁴	72
9	1,1,2,2-tetrachloroethylene	1	6.8×10 ⁻³	35
10	1,1,2,2-tetrachloroethane	1	2.4×10 ⁻²	6
11		entry 10 condition 4	1.4×10 ⁻²	35
12		entry 11 condition 5	3.3×10 ⁻³	15
13	Chloroform	2	9.2×10 ⁻³	80
14		1	4.5×10 ⁻²	340
15		entry 14 condition 3	2.1×10 ⁻²	9000

^aField-effect mobility for the accumulation-mode operation.

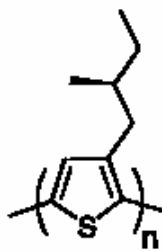
^bOn/off ratio is calculated for enhancement-mode operation only, and it is ten times higher for enhancement-depletion operation.

$\mu_{\text{cast}} > \mu_{\text{spin-coating}}$, probably due to slower evaporation rate

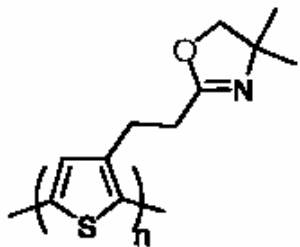
Effect of side chains



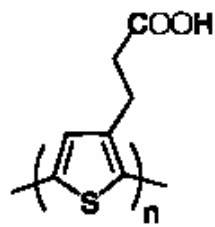
(a)



(b)



(c)



(d)

(a) P3HT has the highest mobility ($\sim 0.1 \text{ cm}^2/\text{Vs}$)

For P3OT ($m = 10$), mobility ($\sim 1 \times 10^{-6} \text{ cm}^2/\text{Vs}$)

From diffraction pattern, the degree of π -overlap does not seem to be affected by longer side chain.

➔ probably due to higher volume fraction of insulating side chains

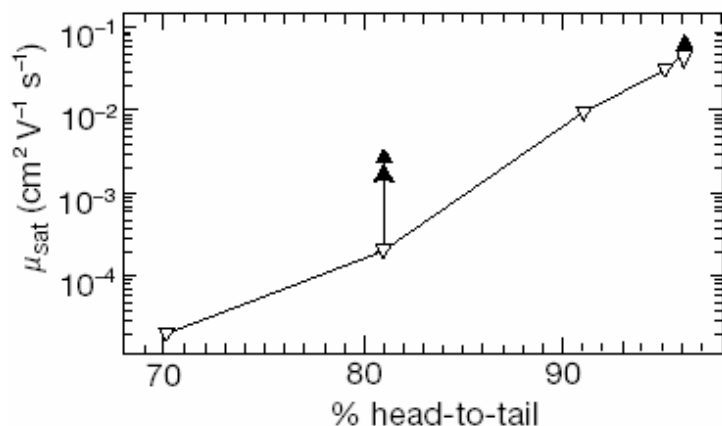
(b) π - π overlap distance increase (4.3 \AA vs 3.9 \AA for P3HT)
($\mu \sim 1 \times 10^{-3} \text{ cm}^2/\text{Vs}$)

(c) and (d)

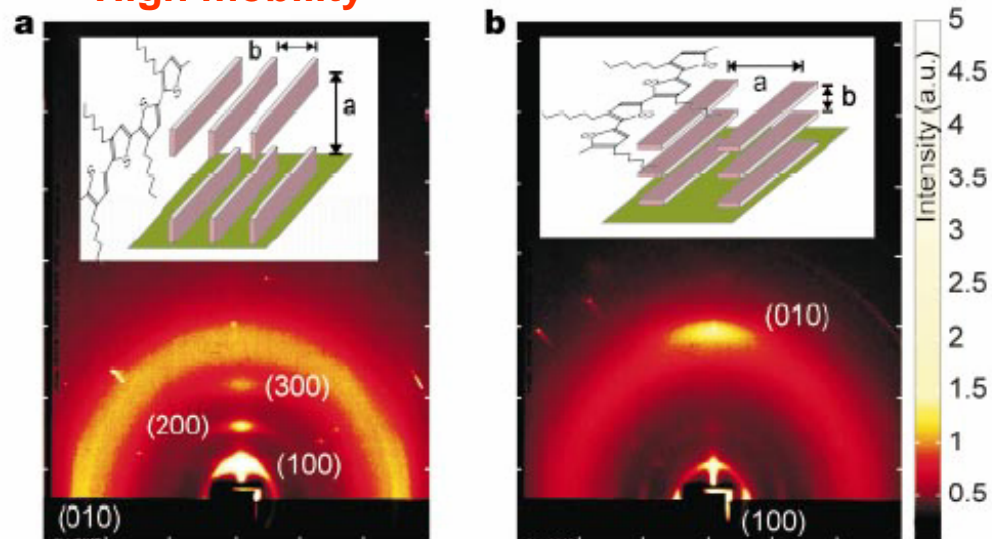
($\mu \sim 1 \times 10^{-5} \text{ cm}^2/\text{Vs}$)

Polymers with bulky side chains are difficult to form highly crystalline films

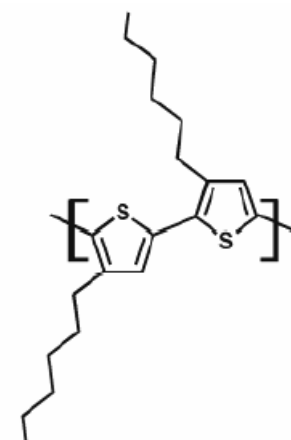
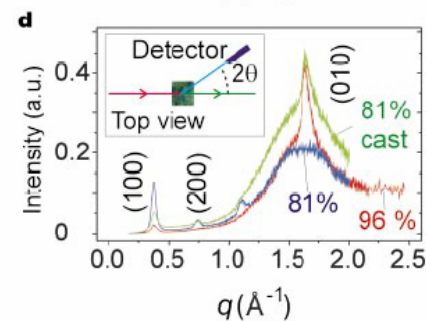
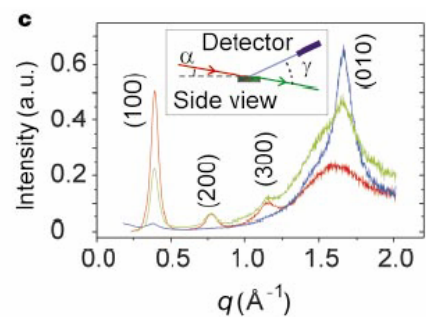
Self-organized polymer TFTs



High mobility



Orientation is also important
High mobility ($\sim 0.1 \text{ cm}^2/\text{Vs}$) was obtained
while the chain edge on



H. Sirringhaus *et al.*, Nature, 401, 685 (1999)

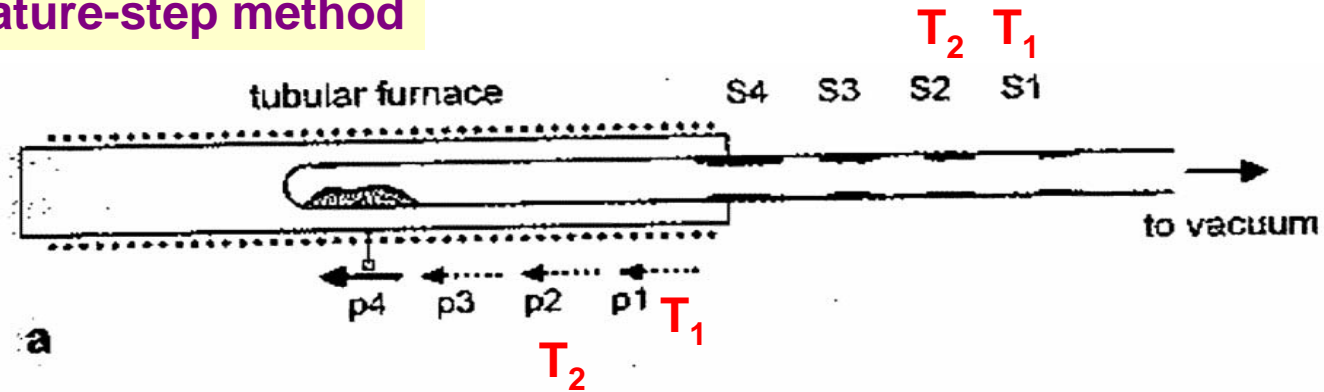
Purification of organic compounds

Methods:

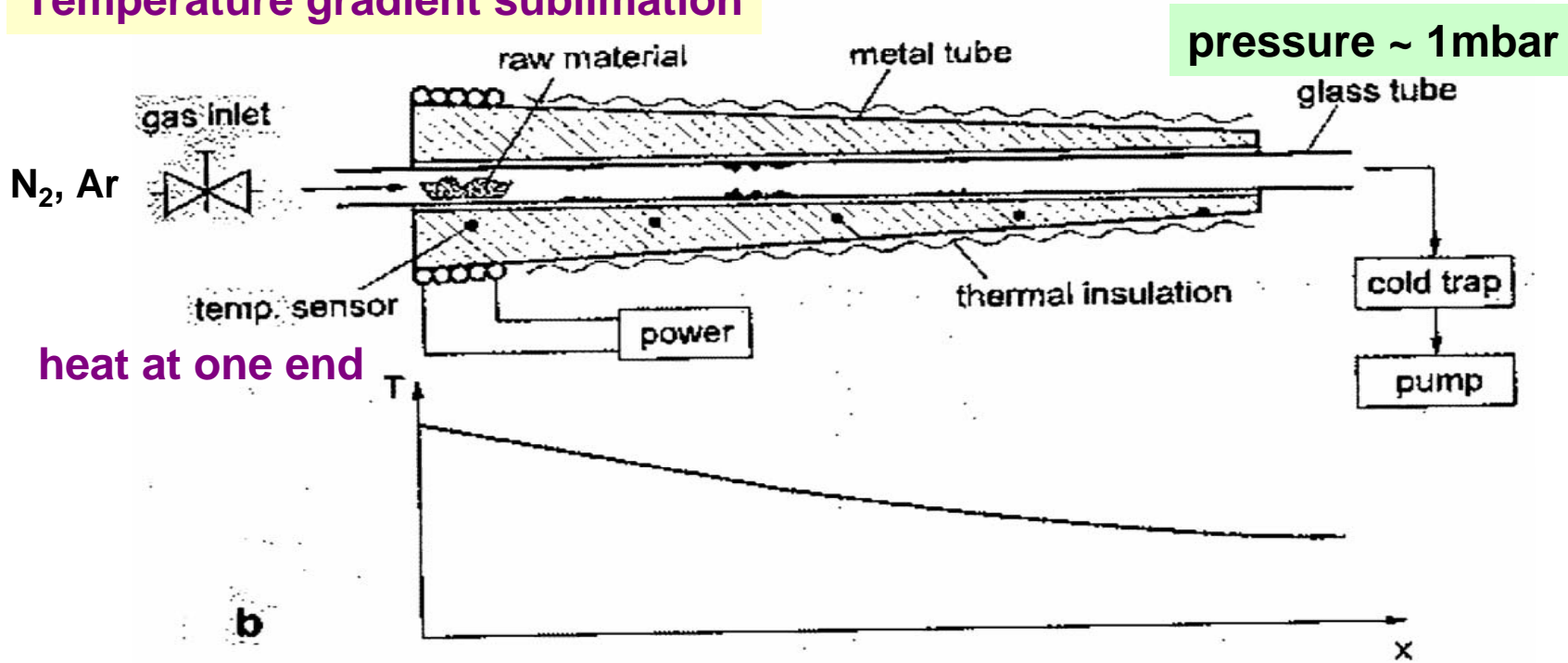
1. Sublimation, applying a temperature-step program or a temperature gradient
2. Recrystallization, from a suitable solvent under slow evaporation or temperature lowering
3. Chromatographic methods, a sequence of phase equilibrium between a solution and an adsorptive surface or thin film.

Sublimation

Temperature-step method



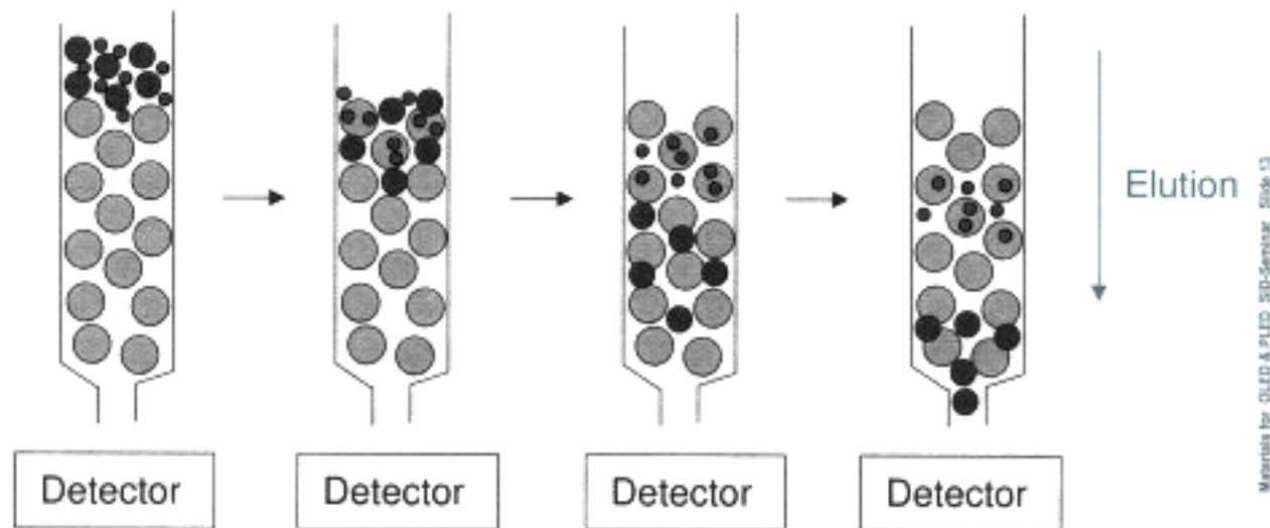
Temperature gradient sublimation



Chromatography

GPC (Gel Permeation Chromatography)

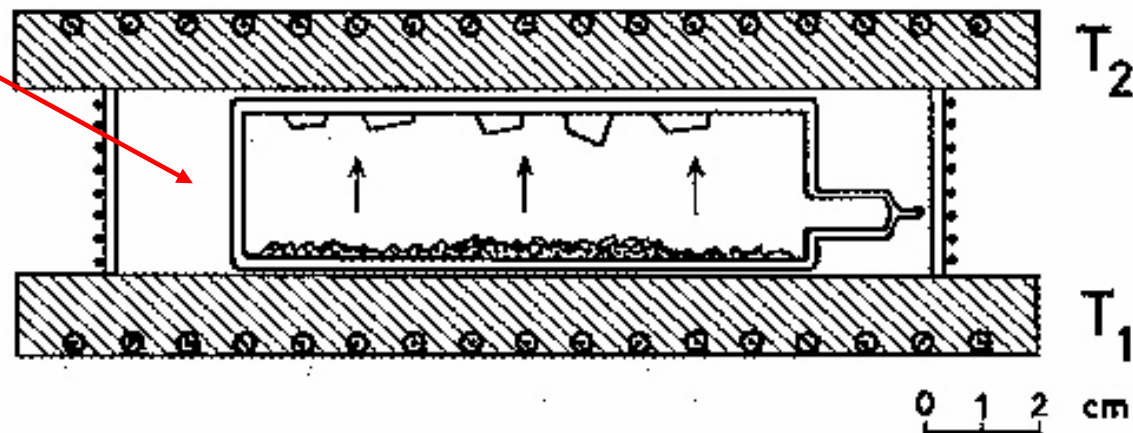
- ▶ Separation by size, not by affinity to stationary material
- ▶ Results strongly depend on shape of polymer (rod-like polymers are usually overestimated)



Growth of single crystals

Plate Sublimation Method

Pyrex glass or
fused silica ampoule



Vertical temperature gradient ($T_2 < T_1$)